



Strong second harmonic generation in LiInX_2 ($X=\text{Se, Te}$) chalcopyrite crystals as explored by first-principles methods



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ABSTRACT

Complex first-order linear optical dispersion and complex second-order non-linear optical dispersion of LiInX_2 ($X = \text{Se, Te}$) chalcopyrite single crystals were calculated in the framework of the density functional theory using the local density approximation (LDA), general gradient approximation (GGA) and modified Becke-Johnson potential. The performed calculations show that these materials possess direct energy band gap ($-\Gamma$) of about 1.58 eV (LDA), 1.85 eV (GGA) and 2.04 eV (mBJ) for LiInSe_2 and 1.76 eV (LDA), 1.92 eV (GGA) and 2.40 eV (mBJ) for LiInTe_2 . Comparison with the experimental data unambiguously indicates that the mBJ succeeds by bringing the calculated gaps in close agreement with the measured ones (2.05 eV - LiInSe_2 and 2.41 eV - LiInTe_2). It has been found that both compounds exhibit negative uniaxial anisotropy and positive birefringence at the static limit. In addition, LiInTe_2 exhibits a higher second harmonic generation (SHG) signal than LiInSe_2 . Moreover, LiInSe_2 (LiInTe_2) exhibits a SHG signal of about seven (nine) times greater than the corresponding experimental value for the well known KTiOPO_4 (KTP) single crystals.

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1. Introduction

In the recent years ternary semiconducting compounds have received much attention [1–3]. The ternary chalcopyrite compounds form a large group of semiconducting materials with diverse optical, electrical and structural properties [4–11]. These compounds appear to be promising candidates for the solar-cells [12], light-emitting diodes [13], nonlinear optics [14] and optical frequency conversion applications in all solid state based tunable laser systems [15]. The ternary chalcopyrite compounds can be grown as thin films and their band gaps can vary from infrared (most of all) to ultraviolet (fewer compounds) spectral regions matching solar spectrum, which makes their applications in solar panels to be quite promising. As an illustration to the last statement, it can be mentioned that the $\text{Cu}(\text{In,Ga})(\text{Se,S})_2$ -based solar elements already question a long dominance of the silicon-based solar panels [16]. Many research works to synthesize novel chalcopyrite materials, both neat and mixed with partial cation or anion

substitution, have been reported. It was recently demonstrated that LiInSe_2 exhibits high potential as a solar cell material [17]. Theoretical calculations for the LiInS_2 , LiInSe_2 and LiInTe_2 chalcopyrites were performed to analyze the lattice dynamics, thermodynamic, elastic properties [18–23].

Lagoun et al. [24] have reported the elastic and piezoelectric properties of LiMX_2 ($M = \text{Ga, In; X} = \text{S, Se}$) crystals. Recently Ma and Brik [25] reported the structural, electronic, and optical properties of LiInSe_2 and LiInTe_2 crystals using the CASTEP code within the local density approximation (LDA) and generalized gradient approximation (GGA). Special emphasis was placed upon the role of the Se and Te anions in formation of the peculiar features of the electronic and optical properties of these compounds.

At the same time it should be stressed out that there is a certain lack of information on the nonlinear optical properties of LiInSe_2 and LiInTe_2 . To fill in this gap, we calculated the nonlinear optical properties of these compounds using the recently modified Becke-Johnson potential (mBJ) based on the calculated band structures within mBJ. Application of mBJ allows for the calculation of the band gaps with accuracy similar to the very expensive GW calculations. It is a local approximation to an atomic “exact-exchange” potential and a screening term. The paper is organized as follows: in the next section we describe the details of calculations and crystal structure

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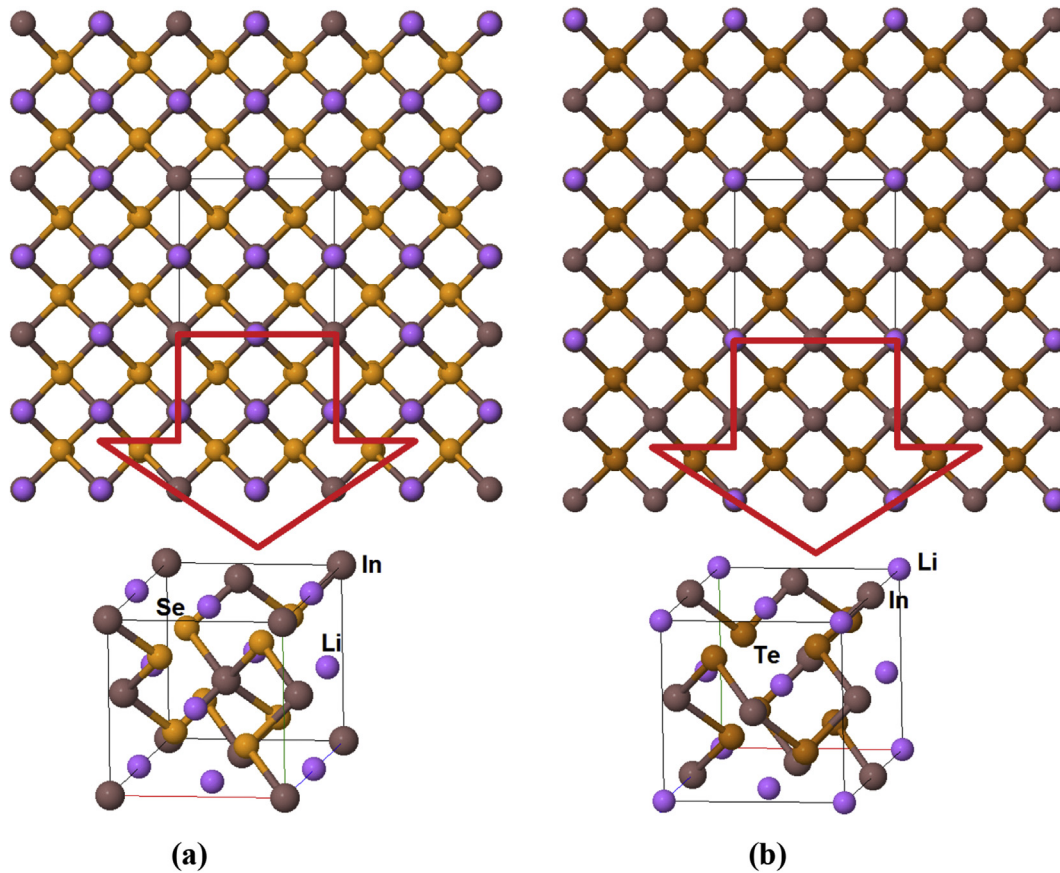


Fig. 1. Crystal structure of LiInSe_2 and LiInTe_2 .

Table 1

Optimized crystal structure of LiInSe_2 in comparison with the experimental data [26,27] and the previous theoretical calculations [25].

Cell parameters (Å)		A		b		c
Exp.		5.807 ^a		5.807 ^a		11.810 ^a
This work		5.823		5.823		11.796
Previous work		6.005 ^b		6.005 ^b		11.875 ^b
		5.835 ^c		5.835 ^c		11.593 ^c
Atomic positions						
Atom	x exp.	x optim.	y exp.	y optim.	z exp.	z optim.
In	0.0	0.0	0.0	0.0	0.0	0.0
Li	0.0	0.0	0.0	0.0	0.5	0.5
Se	0.257	0.244	0.25	0.25	0.125	0.125

^a Ref. [26,27].

^b Ref. Brik (GGA) [25].

^c Ref. Brik (LDA) [25].

Table 2

Optimized crystal structure of LiInTe_2 in comparison with the experimental data [26,27] and the previous theoretical calculations [25].

Cell parameters (Å)		A		b		c
Exp.		6.398 ^a		6.398 ^a		12.460 ^a
This work		6.354		6.354		12.400
Previous work		6.546 ^b		6.546 ^b		12.804 ^b
		6.300 ^c		6.300 ^c		12.482 ^c
Atomic positions						
atom	x exp.	x optim.	y exp.	y optim.	z exp.	z optim.
In	0.0	0.0	0.0	0.0	0.5	0.5
Li	0.0	0.0	0.0	0.0	0.0	0.0
Te	0.241	0.243	0.25	0.25	0.125	0.125

^a Ref. [26,27].

^b Ref. Brik (GGA) [25].

^c Ref. Brik (LDA) [25].

of the studied samples. We proceed with the description of the calculated results and conclude the paper with a summary of the main results obtained.

2. Details of calculations

The ternary chalcopyrite compounds LiInX_2 ($X = \text{Se, Te}$) crystallize in non-centro-symmetric tetragonal space group I-42d with four formula units per one unit cell. The crystal structure of LiInX_2 ($X = \text{Se, Te}$) single crystals is illustrated in Fig. 1. The experimental lattice parameters of LiInSe_2 are $a = b = 5.807 \text{ \AA}$, $c = 11.810 \text{ \AA}$ and $Z = 4$ [26], whereas the same parameters for LiInTe_2 are $a = b = 6.398 \text{ \AA}$, $c = 12.460 \text{ \AA}$ and $Z = 4$ [27]. We have optimized the experimental lattice constant of LiInX_2 ($X = \text{Se, Te}$) using the LDA [28]. Calculations yield the lattice constants $a = b = 5.829 \text{ \AA}$ and $c = 11.738 \text{ \AA}$ for LiInSe_2 , while for LiInTe_2 $a = b = 6.328 \text{ \AA}$ and $c = 12.422 \text{ \AA}$, in good agreement with the experimental values [26,27]. Further the atomic positions were optimized using the generalized gradient approximation (PBE-GGA) [29] by minimizing the forces acting on the atoms. The optimized crystal structure data are listed in Tables 1 and 2 in comparison with the experimental data [26,27] and previous theoretical results [25]. The crystal structure of LiInX_2 ($X = \text{Se, Te}$) single crystals are illustrated in Fig. 1. We have used the relaxed geometrical structure to calculate the band structures and hence the complex linear and nonlinear optical susceptibilities. The all-electron full potential linear augmented plane wave plus local orbitals (FP-LAPW + lo) method in a scalar relativistic version as embodied in the WIEN2k code [30] was employed in these calculations. This is an implementation of the density functional theory (DFT) with different possible approximations for the exchange-correlation (XC) potential. The exchange and correlation potentials were described by the LDA and PBE-GGA approximations, which is based on exchange-correlation energy optimization to calculate the total energy. In addition, we have used the recently modified Becke-

Johnson potential (mBJ) [31], which optimizes the corresponding potential for electronic band structure calculations. The Kohn–Sham equations are solved using a basis of linear APW's. The potential and charge density in the muffin-tin (MT) spheres are expanded in spherical harmonics with $l_{\text{max}} = 8$ and nonspherical components up to $l_{\text{max}} = 6$. In the interstitial region the potential and the charge density are represented by the Fourier series. Self-consistency of the calculated results is obtained using 500 k points in the irreducible Brillouin zone (IBZ). We have calculated the linear optical susceptibilities using 800 k points and the nonlinear optical susceptibilities using 1500 k points in the IBZ.

The non-centro-symmetric tetragonal (I-42d) symmetry allows three non-zero components of the second-order dielectric (optical) tensor corresponding to the electric field \vec{E} being directed along the **a**, **b**, and **c**-crystallographic axes. These are $\epsilon^{xx}(\omega) = \epsilon^{yy}(\omega) = \epsilon_2^x(\omega)$ and $\epsilon^{zz}(\omega) = \epsilon_2^z(\omega)$. The imaginary part $\epsilon_2^x(\omega)$ and $\epsilon_2^z(\omega)$ of the principal complex tensor components completely defines the linear optical susceptibilities. The imaginary part can be obtained using the following expression [32]:

$$\epsilon_2^{ij}(\omega) = \frac{8\pi^2 \hbar^2 e^2}{m^2 V} \sum_k \sum_{cv} (f_c - f_v) \frac{p_{cv}^i(k) p_{vc}^j(k)}{E_{vc}^2} \delta[E_c(k) - E_v(k) - \hbar\omega], \quad (1)$$

where m , e and \hbar are the electron mass, charge and Planck's constant, respectively. f_c and f_v represent the Fermi distributions of the conduction and valence bands, respectively. The term $p_{cv}^i(k)$ denotes the momentum matrix element transition from the energy level c of the conduction band to the level v of the valence band at certain \mathbf{k} -point in the BZ and V is the unit cell volume. The real part of the dielectric function is obtained afterwards using Kramer–Kronig relationship [33].

Furthermore, we have calculated the complex second-order

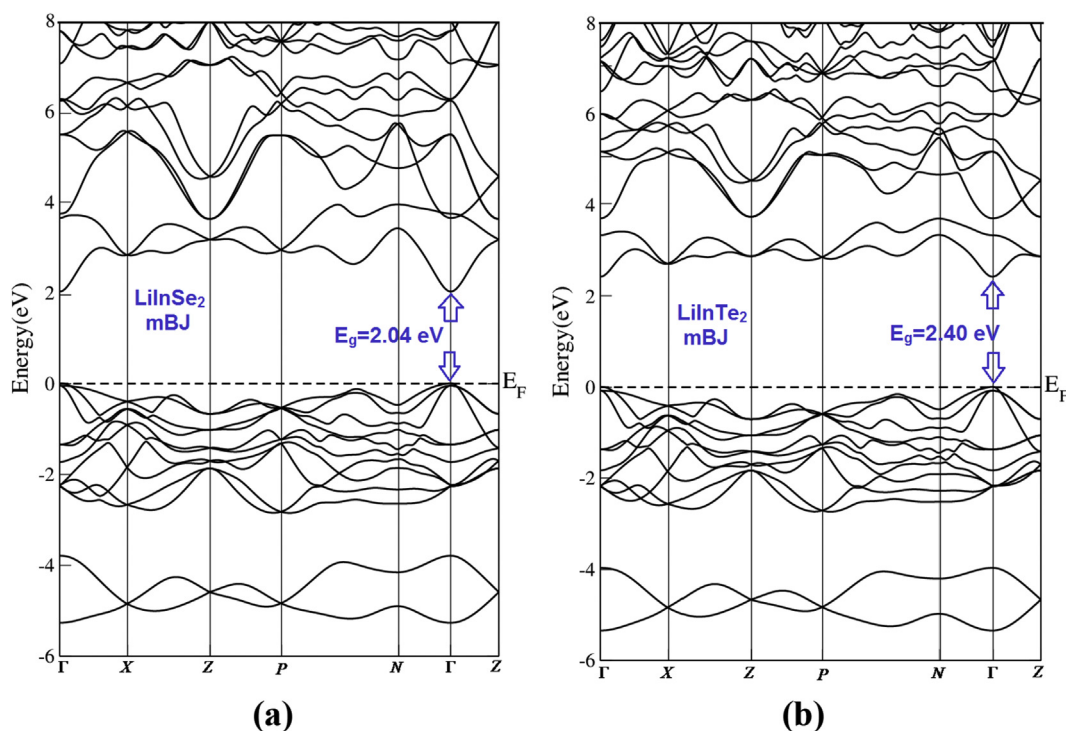


Fig. 2. Calculated electronic band structure using *mBJ* for: (a) LiInSe_2 ; (b) LiInTe_2 .

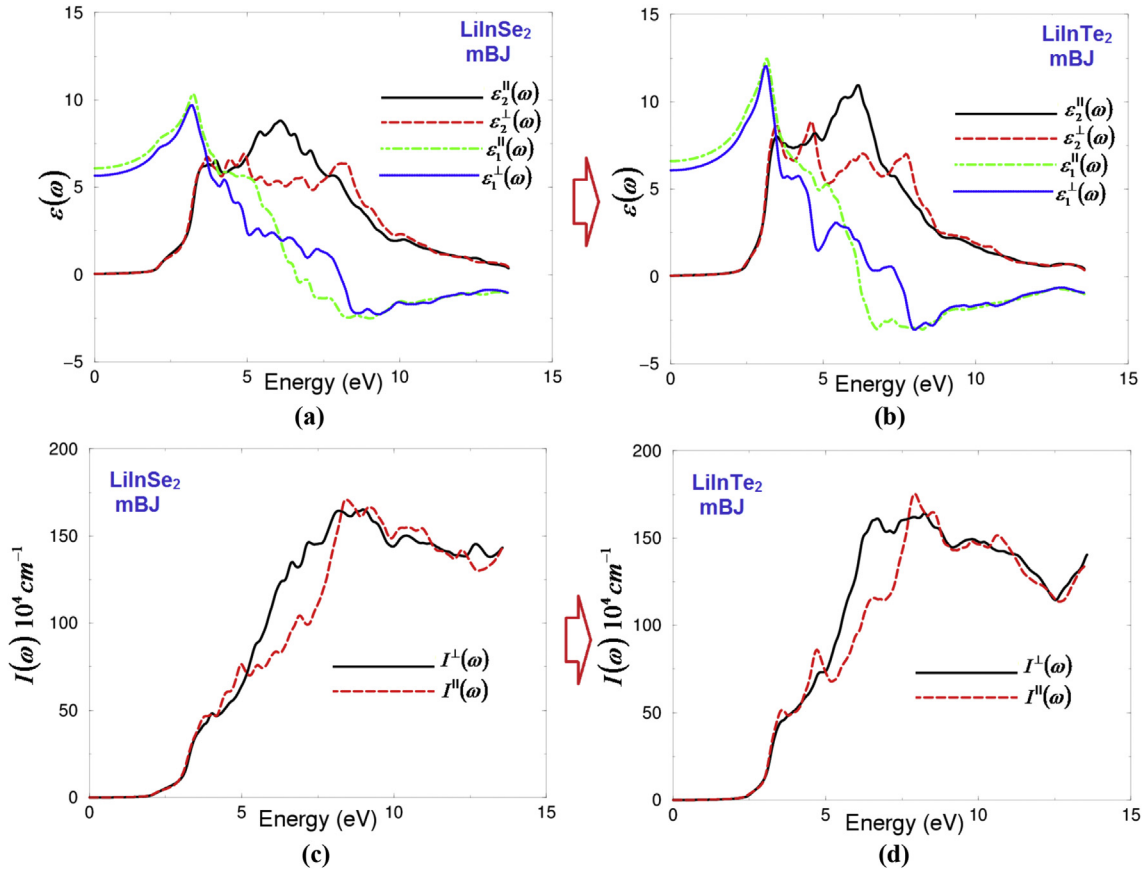


Fig. 3. (a) Calculated $\varepsilon_2^\perp(\omega)$ dispersion (dark solid curve-black), $\varepsilon_2^\parallel(\omega)$ (light dashed curve-red) along with calculated $\varepsilon_1^\perp(\omega)$ (light dotted dashed curve -green), $\varepsilon_1^\parallel(\omega)$ (light dotted curve -blue) for LiInSe₂ compound using mBJ; (b) Calculated $\varepsilon_2^\perp(\omega)$ dispersion (dark solid curve-black), $\varepsilon_2^\parallel(\omega)$ (light dashed curve-red) along with calculated $\varepsilon_1^\perp(\omega)$ (light dotted dashed curve -green), $\varepsilon_1^\parallel(\omega)$ (light dotted curve -blue) for LiInTe₂ compound using mBJ; (c) Calculated $I^\perp(\omega)$ (dark solid curve-black), $I^\parallel(\omega)$ (light dashed curve-red) for LiInSe₂ compound using mBJ; (d) Calculated $I^\perp(\omega)$ (dark solid curve-black), $I^\parallel(\omega)$ (light dashed curve-red) for LiInTe₂ compound using mBJ; (e) Calculated $n^\perp(\omega)$ (dark solid curve-black), $n^\parallel(\omega)$ (light dashed curve-red) for LiInSe₂ compound using mBJ; (f) Calculated $n^\perp(\omega)$ (dark solid curve-black), $n^\parallel(\omega)$ (light dashed curve-red) for LiInTe₂ compound using mBJ; (g) Calculated $n^\perp(\omega)$ (dark solid curve-black), $n^\parallel(\omega)$ (light dashed curve-red) for LiInSe₂ compound using mBJ; (h) Calculated $n^\perp(\omega)$ (dark solid curve-black), $n^\parallel(\omega)$ (light dashed curve-red) for LiInTe₂ compound using mBJ. (i) Calculated $\Delta n(\omega)$ (dark solid curve-black) for LiInSe₂, and (light dashed curve-red) for LiInTe₂. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

nonlinear optical susceptibility tensor $\chi_{ijk}^{(2)}(-2\omega; \omega, \omega)$. These can be generally written as [34–37]:

$$\chi_{\text{inter}}^{ijk}(-2\omega; \omega, \omega) = \frac{e^3}{\hbar^2} \sum_{nml} \int \frac{d\vec{k}}{4\pi^3} \frac{\vec{r}_{nm}^i \{ \vec{r}_{ml}^j \vec{r}_{ln}^k \}}{(\omega_{ln} - \omega_{ml})} \left\{ \frac{2f_{nm}}{(\omega_{mn} - 2\omega)} + \frac{f_{ml}}{(\omega_{ml} - \omega)} + \frac{f_{ln}}{(\omega_{ln} - \omega)} \right\} \quad (2)$$

$$\chi_{\text{intra}}^{ijk}(-2\omega; \omega, \omega) = \frac{e^3}{\hbar^2} \int \frac{d\vec{k}}{4\pi^3} \left[\sum_{nml} \omega_{nm} \vec{r}_{nm}^i \{ \vec{r}_{ml}^j \vec{r}_{ln}^k \} \times \left\{ \frac{f_{nl}}{\omega_{ln}^2(\omega_{ln} - \omega)} - \frac{f_{lm}}{\omega_{ml}^2(\omega_{ml} - \omega)} \right\} - 8i \right. \\ \times \sum_{nm} \frac{f_{nm} \vec{r}_{nm}^i \{ \Delta_{mn}^j \vec{r}_{nm}^k \}}{\omega_{mn}^2(\omega_{mn} - 2\omega)} + 2 \\ \left. \times \sum_{nml} \frac{f_{nm} \vec{r}_{nm}^i \{ \vec{r}_{ml}^j \vec{r}_{ln}^k \} (\omega_{ml} - \omega_{ln})}{\omega_{mn}^2(\omega_{mn} - 2\omega)} \right] \quad (3)$$

$$\chi_{\text{mod}}^{ijk}(-2\omega; \omega, \omega) = \frac{e^3}{2\hbar^2} \int \frac{d\vec{k}}{4\pi^3} \left[\times \sum_{nml} \frac{f_{nm}}{\omega_{mn}^2(\omega_{mn} - \omega)} \{ \omega_{nl} \vec{r}_{lm}^i \{ \vec{r}_{mn}^j \vec{r}_{nl}^k \} - \omega_{lm} \vec{r}_{nl}^i \{ \vec{r}_{lm}^j \vec{r}_{mn}^k \} \} - i \right. \\ \left. \times \sum_{nm} \frac{f_{nm} \vec{r}_{nm}^i \{ \vec{r}_{mn}^j \Delta_{mn}^k \}}{\omega_{mn}^2(\omega_{mn} - \omega)} \right] \quad (4)$$

From these formulae we can notice that there are three major contributions to $\chi_{ijk}^{(2)}(-2\omega; \omega, \omega)$: the inter-band transitions $\chi_{\text{inter}}^{ijk}(-2\omega; \omega, \omega)$, the intra-band transitions $\chi_{\text{intra}}^{ijk}(-2\omega; \omega, \omega)$ and the modulation of inter-band terms by intra-band terms $\chi_{\text{mod}}^{ijk}(-2\omega; \omega, \omega)$, where $n \neq m \neq l$. Here n denotes the valence states, m the conduction states and l denotes all states ($l \neq m, n$). There are two kinds of transitions that can take place, one of them denoted as vc' , involves one valence band (v) and two conduction bands (c and c'), and the second transition denoted as $vv'c$, involves two valence bands (v and v') and one conduction band (c). The symbols

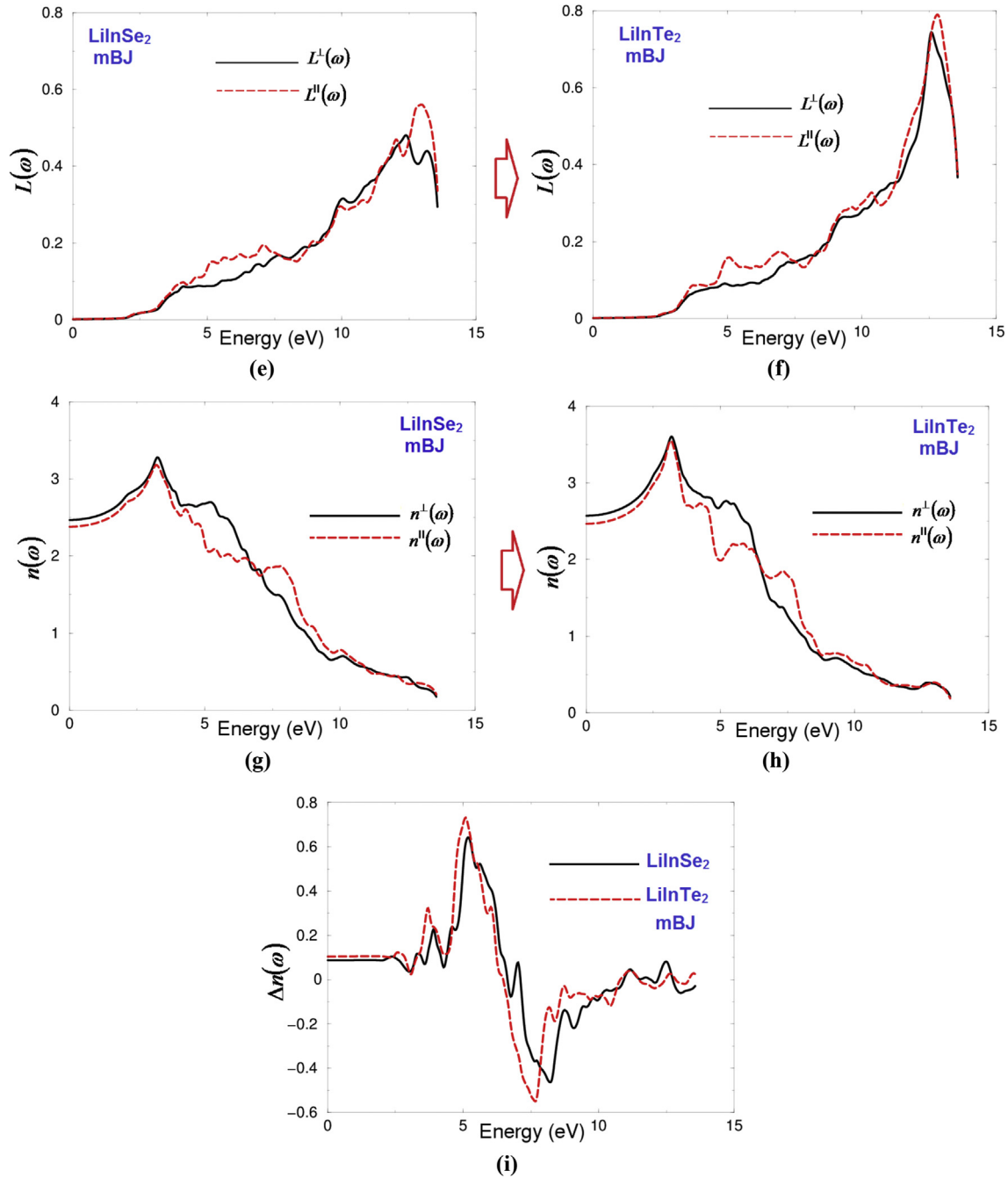


Fig. 3. (Continued)

are defined as $\Delta_{nm}^i(\vec{k}) = \vartheta_{nm}^i(\vec{k}) - \vartheta_{nm}^i(\vec{k})$ with ϑ_{nm}^i being the i component of the electron velocity given as $\vartheta_{nm}^i(\vec{k}) = i\omega_{nm}(\vec{k})r_{nm}^i(\vec{k})$ and $\{r_{nm}^i(\vec{k})r_{ml}^j(\vec{k})\} = \frac{1}{2}(r_{nm}^i(\vec{k})r_{ml}^j(\vec{k}) + r_{nm}^j(\vec{k})r_{ml}^i(\vec{k}))$. The position matrix elements between band states n and m , $r_{nm}^i(\vec{k})$, are calculated from the momentum matrix element P_{nm}^i using the relation

[38]: $r_{nm}^i(\vec{k}) = \frac{P_{nm}^i(\vec{k})}{i m \omega_{nm}(\vec{k})}$, with the energy difference between the

states n and m given by $\hbar\omega_{nm} = \hbar(\omega_n - \omega_m)$. $f_{nm} = f_n - f_m$ is the difference of the Fermi distribution functions. The i , j and k entries correspond to the Cartesian indices. It has been demonstrated by Aspnes [39] that only the one-electron virtual transitions

(transitions between one valence band state and two conduction band states, vcc') give a significant contribution to the second-order tensor. We ignore the virtual-hole contribution (transitions between two valence band states and one conduction band state, $vv'c$), because it was found to be negative and more than an order of magnitude smaller than the virtual-electron contribution for this compound.

3. Results and discussion

3.1. Salient features of the electronic band structures

To ascertain the influence of substituting Se by Te on the electronic properties of LiInX₂ ($X = \text{Se, Te}$) single crystals, we have

Table 3

The calculated energy band gap of LiInSe₂ and LiInTe₂ in comparison with the experimental data [26,27] and the previous calculation [25] $\epsilon_1^{\perp}(0)\epsilon_1^{\parallel}(0)\epsilon_1^{\text{average}}(0)$, $\delta\epsilon$, $\omega_p^{\perp}(\omega)$, $\omega_p^{\parallel}(\omega)n^{\perp}(0)$, $n^{\parallel}(0)$ and $\Delta n(0)$. In this work the parameters are calculated within mBJ.

	LiInSe ₂	LiInTe ₂
Eg (eV)	2.04 ^a , 2.05 ^a , 1.684 ^c , 1.615 ^d	2.40 ^a , 2.41 ^b , 1.313 ^c , 1.513 ^d
$\epsilon_1^{\perp}(0)$	6.091	6.609
$\epsilon_1^{\parallel}(0)$	5.666	6.081
$\epsilon_1^{\text{tot}}(0)$	11.757	12.690
$\delta\epsilon$	-0.036	-0.041
$\omega_p^{\perp}(\omega)$	6.598	6.190
$\omega_p^{\parallel}(\omega)$	8.040	7.442
$n^{\perp}(0)$	2.468	2.570
$n^{\parallel}(0)$	2.380	2.466
$\Delta n(0)$	0.088	0.104

^aThis work.

^b Ref. [32].

^c Ref. [33].

^d Ref. [31] GGA.

^e Ref. [31] LDA.

calculated the electronic band structures using mBJ as shown in Fig. 2(a) and (b). It has been noticed that these materials possess

direct energy band gap (Γ - Γ) of about 1.58 eV (LDA), 1.85 eV (GGA) and 2.04 eV (mBJ) for LiInSe₂, while the same band gaps are 1.76 eV (LDA), 1.92 eV (GGA) and 2.40 eV (mBJ) for LiInTe₂. It is clear that mBJ ensures much better agreement with the experimental band gaps (2.05 eV -LiInSe₂ and 2.41 eV -LiInTe₂) [26,27].

3.2. Complex first-order linear optical dispersion

Using Eq. (1) and following the dipolar selection rule which states that only those transitions, in which the angular momentum quantum number l is changed by unity ($\Delta l = \pm 1$) are allowed, we have calculated the imaginary and real parts of the optical dielectric functions for the LiInX₂ (X = Se, Te) single crystals using mBJ; the obtained results are shown in Fig. 3 (a) and 3 (b). It has been found that substituting Se by Te causes a shift of the whole spectral structure towards higher energies by around 0.36 eV with increasing the magnitude of the spectral structures. This result implies that the optical band gap is increased when we move from Se to Te in concordance with the calculated electronic band structure. The first absorption edges are located at 2.05 (2.41) eV for LiInSe₂ (LiInTe₂). These are due to optical transition between the Li-2s, In-5s, Se-4p (Te-5p) valence bands to the In-5s/5p, Se4s/4p (Te5s/5p) conduction bands. We observed that the spectral structure of $\epsilon_2^{\perp}(\omega)$ for LiInSe₂ exhibits one main peak situated between

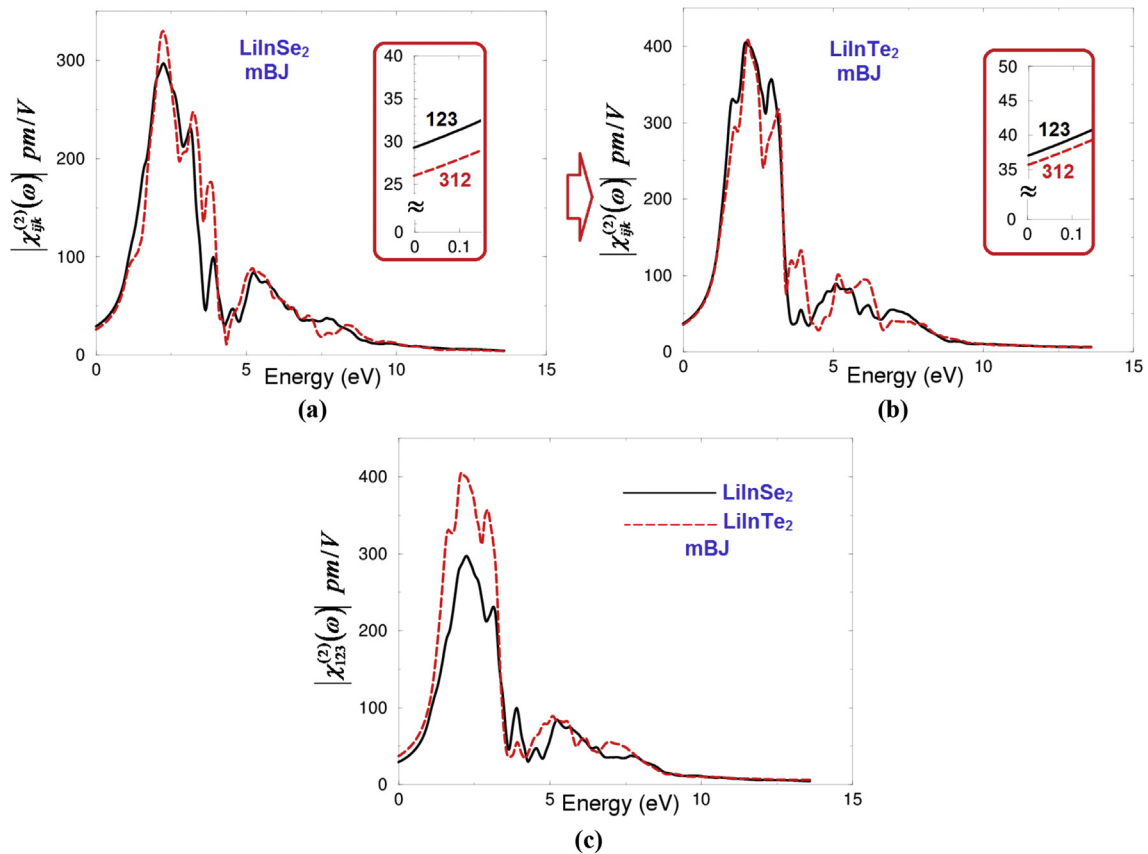


Fig. 4. (a) Calculated $|\chi_{ijk}^{(2)}(\omega)|$ for the two tensor components of LiInSe₂; (b) Calculated $|\chi_{ijk}^{(2)}(\omega)|$ for the five tensor components of LiInTe₂; (c) Calculated Imaginary $\chi_{123}^{(2)}(\omega)$ (dark solid curve-black) and real $\chi_{123}^{(2)}(\omega)$ (light dashed curve-red) spectra of LiInSe₂; (d) Calculated Imaginary $\chi_{333}^{(2)}(\omega)$ (dark solid curve-black) and real $\chi_{333}^{(2)}(\omega)$ (light dashed curve-red) spectra of LiInTe₂; (e) Calculated total $\text{Im}\chi_{123}^{(2)}(\omega)$ spectrum (dark solid curve-black) along with the intra $(2\omega)/(1\omega)$ (light solid curve-blue)/(light dashed dotted curve-cyan) and inter $(2\omega)/(1\omega)$ (light long dashed curve-red)/(light dotted curve-green) -band contributions of LiInSe₂, here all $\text{Im}\chi_{123}^{(2)}(\omega)$ are multiplied by 10^{-7} , in esu units; (f) Calculated total $\text{Im}\chi_{123}^{(2)}(\omega)$ spectrum (dark solid curve-black) along with the intra $(2\omega)/(1\omega)$ (light solid curve-blue)/(light dashed dotted curve-cyan) and inter $(2\omega)/(1\omega)$ (light long dashed curve-red)/(light dotted curve-green) -band contributions of LiInTe₂, here all $\text{Im}\chi_{123}^{(2)}(\omega)$ are multiplied by 10^{-7} , in esu units; (g) -upper panel- Calculated $|\chi_{123}^{(2)}(\omega)|$ (dark solid curve-black); -lower panel- Calculated $\epsilon_2^{xx}(\omega)$ (dark solid curve-black); Calculated $\epsilon_2^{xx}(\omega)/2$ (dark dashed curve-red) of LiInSe₂; (h) -upper panel- Calculated $|\chi_{123}^{(2)}(\omega)|$ (dark solid curve-black); -lower panel- Calculated $\epsilon_2^{xx}(\omega)$ (dark solid curve-black); Calculated $\epsilon_2^{xx}(\omega)/2$ (dark dashed curve-red) of LiInTe₂. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

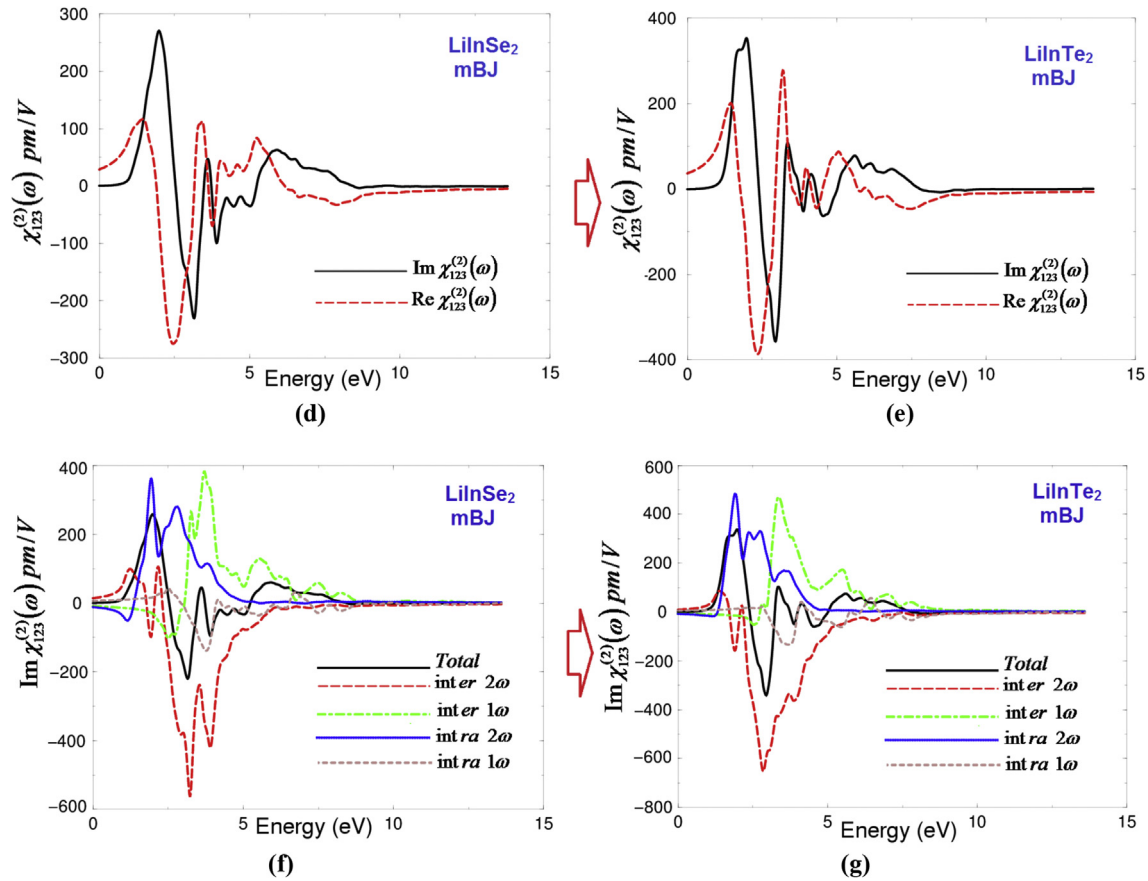


Fig. 4. (Continued)

two humps, whereas $\epsilon_2^H(\omega)$ show multi humps in the main structure. At the same time, the $\epsilon_2^L(\omega)$ for LiInTe₂ shows one main peak and two humps located at the left side of the main peak, whereas $\epsilon_2^H(\omega)$ exhibit multi peaks in the main structure. The main spectral structures are due to the optical transitions between the Li-2s, In5s/5p, Se-4p (Te-5p) valence bands to Li-2s, In-5s/5p, Se-4s/4p (Te-5s/5p) conduction bands. The strength of the main peaks could be explained by the fact that $\epsilon_2(\omega)$ scales as $1/\omega^2$.

From the calculated real part $\epsilon_1^L(\omega)$ and $\epsilon_1^H(\omega)$ we have obtained the vanishing frequency value of the dielectric function, which defines the static electronic dielectric constant $\epsilon_1^L(0)$ and $\epsilon_1^H(0)$. These values are listed in Table 3. Using these values we can estimate the energy gap from the Penn model [40], $\epsilon(0) \approx 1 + (\hbar\omega_p/E_g)^2$. It is clear that $\epsilon(0)$ is inversely proportional to E_g therefore, smaller $\epsilon(0)$ would correspond to larger E_g . Following Table 3 it has been found that the value of $\epsilon(0)$ for LiInSe₂ > LiInTe₂. This is another evidence that moving from Se to Te in LiInX₂ increases the band gap. Furthermore, the $\epsilon(0)$ values help to estimate the uniaxial anisotropy as listed in Table 3. From the real parts of the dielectric function we can obtain the plasmon oscillations $\omega_p^L(\omega)$ and $\omega_p^H(\omega)$ (Table 3), which occur at energy where $\epsilon_1(\omega)$ crosses zero. It is associated with the existence of plasma oscillations (plasmons). With the aid of existing information about $\epsilon_1(\omega)$ and $\epsilon_2(\omega)$, the other optical properties can be calculated. The absorption coefficients of LiInSe₂ and LiInTe₂ as illustrated in Fig. 3(c) and (d) indicate that the absorption edges occur at 2.04 eV and 2.40 eV in concordance with our observation from $\epsilon_2(\omega)$ (Fig. 3(a) and (b)) that the energy band gap enlargement occurs when we substituting Se by Te. It has been noticed that LiInSe₂ and LiInTe₂ possess relatively wide optical transparency region.

Fig. 3(e) and (f) show the energy loss function $L(\omega)$ of LiInSe₂ and LiInTe₂, the lossless region of LiInSe₂ is confined between 10.0 eV and 13.0 eV, whereas it is between 11.0 eV and 14.0 eV for LiInTe₂. There are some other features in the $L(\omega)$ behavior, in addition to the plasmon peak, associated with the interband transitions. The most intensive feature is the plasmon peak, it occurs at energy region where $\epsilon_1(\omega)$ goes to zero. In Fig. 3(g) and (h) the calculated refractive indices of LiInSe₂ and LiInTe₂ are plotted, the values of $n^L(0)$ and $n^H(0)$ were calculated and presented in Table 3. From the calculated refractive indices we can estimate the birefringence from the following expression $\Delta n(\omega) = n_e(\omega) - n_o(\omega)$. It has been found that both compounds exhibit positive birefringence at the static limit as shown in Table 3. The birefringence is necessary to fulfill the phase-matching condition. The dispersion of the birefringence of the two compounds is illustrated in Fig. 3(i).

3.3. Complex second-order non-linear optical dispersion

It has been reported that the materials, which show a considerable anisotropy in the linear optical susceptibilities, favor an enhanced phase matching conditions for the second harmonic generation (SHG) and optical parametric oscillation (OPO). Also the materials with a wide band gap possess high laser damage threshold [41,42]. It is clear that from the calculated linear optical properties both LiInSe₂ and LiInTe₂ exhibit a considerable anisotropy between the two optical components. As a result, these materials are expected to produce strong SHG. Furthermore the calculated electronic band structure reveals that these compounds possess a wide energy band gap and hence they exhibit high laser damage threshold. In this section we turn our concern to calculate the

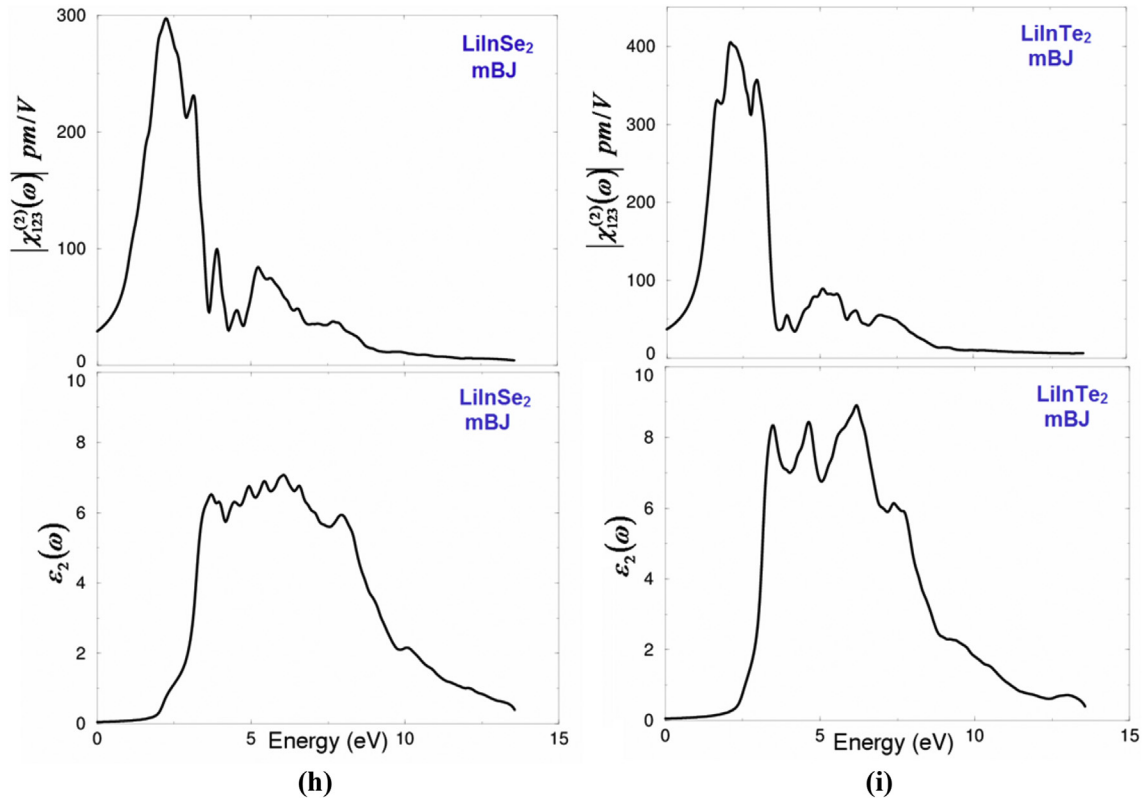


Fig. 4. (Continued)

Table 4
Calculated $|\chi_{ijk}^{(2)}(\omega)|$ of LiInSe₂, in pm/V at static limit and at $\lambda = 1064$ nm, in comparison with the experimental value of the well known KTiOPO₄ (KTP) single crystals which exhibits a SHG value of about 16.9 [43], 13.7 [44], 15.4 ± 0.2 [45], 14.6 ± 1.0 [46], 17.4 ± 1.7 [47], 16.9 ± 3.3 [48], 16.9 ± 1.7 [49], 10.6 ± 7.5 [50], 16.75 [51] and 16.65 [51] at $\lambda = 1064$. Where $1 \text{ pm/V} = 2.387 \times 10^{-9}$ esu.

LiInSe ₂				
Tensor components	$\chi_{ijk}^{(2)}(0)$	Theory $d_{ijk} = 0.5\chi_{ijk}^{(2)}(\omega)$ $d_{14} = 14.588$	$\chi_{ijk}^{(2)}(\omega)$ at $\lambda = 1064$	Theory $d_{ijk} = 0.5\chi_{ijk}^{(2)}(\omega)$ $d_{14} = 55.732$
$ \chi_{123}^{(2)}(\omega) $	29.177		111.465	
$ \chi_{312}^{(2)}(\omega) $	26.002	$d_{36} = 13.001$	95.861	$d_{36} = 47.930$

complex second-order non-linear optical properties of LiInSe₂ and LiInTe₂ single crystals.

Since both LiInSe₂ and LiInTe₂ single crystals are crystallized in tetragonal symmetry, only two non-zero tensor components $\chi_{123}^{(2)}(-2\omega; \omega; \omega)$ and $\chi_{312}^{(2)}(-2\omega; \omega; \omega)$ are allowed. The calculated absolute values of $|\chi_{123}^{(2)}(-2\omega; \omega; \omega)|$ and $|\chi_{312}^{(2)}(-2\omega; \omega; \omega)|$ for both compounds reveal that the $|\chi_{123}^{(2)}(-2\omega; \omega; \omega)|$ component is the dominant one (Fig. 4(a) and (b)). It has been found that LiInTe₂ exhibits higher SHG than that one obtained from LiInSe₂. These values are listed in Tables 4 and 5. We also found that LiInSe₂ (LiInTe₂) exhibit a SHG of about seven (nine) times than that the experimental value of the well known KTiOPO₄ (KTP) single

crystals. We would like to mention that the experimental value of the SHG signal for the well known KTiOPO₄ (KTP) single crystals of about 16.9 [43], 13.7 [44], 15.4 ± 0.2 [45], 14.6 ± 1.0 [46], 17.4 ± 1.7 [47], 16.9 ± 3.3 [48], 16.9 ± 1.7 [49], 10.6 ± 7.5 [50], 16.75 [51] and 16.65 [51].

Further detailed information regarding the origin of the strong SHG for the two compounds can be obtained from analyzing the real and imaginary parts of the dominant component $\chi_{123}^{(2)}(-2\omega; \omega; \omega)$ as shown in Fig. 4(c) and (d). It has been found that the 2ω terms begin to oscillate at the half value of the energy band gap, whereas the ω terms oscillate at the exact value of the fundamental gap to be added to 2ω terms. Both of 2ω and ω terms

Table 5
Calculated $|\chi_{ijk}^{(2)}(\omega)|$ of LiInTe₂ in pm/V at static limit and at $\lambda = 1064$ nm, in comparison with the experimental value of the well known KTiOPO₄ (KTP) single crystals which exhibits a SHG value of about 16.9 [43], 13.7 [44], 15.4 ± 0.2 [45], 14.6 ± 1.0 [46], 17.4 ± 1.7 [47], 16.9 ± 3.3 [48], 16.9 ± 1.7 [49], 10.6 ± 7.5 [50], 16.75 [51] and 16.65 [51] at $\lambda = 1064$. Where $1 \text{ pm/V} = 2.387 \times 10^{-9}$ esu.

LiInTe ₂				
Tensor components	$\chi_{ijk}^{(2)}(0)$	Theory $d_{ijk} = 0.5\chi_{ijk}^{(2)}(\omega)$ $d_{14} = 18.529$	$\chi_{ijk}^{(2)}(\omega)$ at $\lambda = 1064$	Theory $d_{ijk} = 0.5\chi_{ijk}^{(2)}(\omega)$ $d_{14} = 70.365$
$ \chi_{123}^{(2)}(\omega) $	37.058		140.73	
$ \chi_{312}^{(2)}(\omega) $	35.646	$d_{36} = 17.823$	136.917	$d_{36} = 68.458$

are further separate to inter- and intra band transitions as illustrated in Fig. 4(e) and (f).

To identify the origin of the spectral peaks as caused by the $2\omega/\omega$ inter-intra-band contributions, we have associated the spectral structures of the dominant component to the absorptive part of the optical dielectric function $\varepsilon_2(\omega)$ as a function of both $\omega/2$ and ω as shown in Fig. 4(g) and (h). The first structure in $|\chi_{123}^{(2)}(\omega)|$ between 1.02 and 2.05 eV (1.20–2.41 eV) for LiInSe_2 (LiInTe_2) is mainly originated from the 2ω resonance. The second structure between 2.05 and 5.0 eV (2.41–5.0 eV) is associated with interference between 2ω and ω resonances (the absorption edge of $\varepsilon_2(\omega)$). The last spectral structure from 5.0 eV and above is mainly due to ω resonance and is associated with the second structure in $\varepsilon_2(\omega)$.

4. Conclusions

We have optimized the experimental lattice parameters of two LiInX_2 ($X = \text{Se, Te}$) chalcopyrite crystals using the all-electron full potential linear augmented plane wave plus local orbitals (FP-LAPW + lo) method in a scalar relativistic version as embodied in the WIEN2k code. The exchange and correlation potentials were described by local density approximation (LDA), the generalized gradient approximation (PBE-GGA) and the recently modified Becke-Johnson potential (mBJ). The complex first-order linear optical dispersion and complex second-order non-linear optical dispersion of the LiInX_2 ($X = \text{Se, Te}$) single crystals were calculated based on density functional theory. Calculations show that these materials possess direct energy band gap (Γ – Γ) of about 1.58 eV (LDA), 1.85 eV (GGA) and 2.04 eV (mBJ) for LiInSe_2 . While it is 1.76 eV (LDA), 1.92 eV (GGA) and 2.40 eV (mBJ) for LiInTe_2 . The mBJ method is better to be used for getting the calculated band gaps to be closer to the experimental ones (2.05 eV - LiInSe_2 and 2.41 eV - LiInTe_2). It has been found that both compounds exhibit negative uniaxial anisotropy and positive birefringence at the static limit. To identify the origin of the spectral peaks as caused by $2\omega/\omega$ inter-intra-band contributions, we have associated the spectral structures of the dominant component to the absorptive part of the optical dielectric function $\varepsilon_2(\omega)$ as a function of both $\omega/2$ and ω . It has been found that LiInTe_2 exhibit a higher SHG signal than that obtained from LiInSe_2 . We found that LiInSe_2 (LiInTe_2) exhibit a SHG of about seven (nine) times than that the experimental value of the well known KTiOPO_4 (KTP) single crystals.

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