



Novel ternary semiconductor CdLa₂X₄ (X=S or Se) single crystal with efficient second harmonic generation in the visible spectral range



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ABSTRACT

Comprehensive *ab-initio* calculations are performed to investigate the suitability of the non-centro-symmetric CdLa₂S₄ and CdLa₂Se₄ for using as nonlinear optical crystals in visible spectral range. The calculations reveal the direct band gap nature of both compounds with large absorption coefficient (10⁴ cm⁻¹). The absorption edges of CdLa₂S₄ and CdLa₂Se₄ occur at λ = 579.3 nm and λ = 670.1 nm, and the optical band gaps are estimated to be 2.14 eV and 1.85 eV for CdLa₂S₄ and CdLa₂Se₄, respectively. CdLa₂S₄ exhibits SHG of about two times of the well-known KH₂PO₄ (KDP) single crystals which exhibits a nonlinear coefficient of 0.39 p.m./V. Whereas CdLa₂Se₄ exhibits SHG very close to the experimental value of that for the well-known KTiOPO₄ (KTP) which exhibits a nonlinear coefficient of about 14.6 ± 1.0.

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1. Introduction

In recent years, tremendous research work have been done for the development of different varieties of novel inorganic and organic nonlinear optical materials (NLO) [1]. The NLO materials exhibit promising applications for second harmonic generation (SHG), sum or difference frequency mixing, optical parametric oscillation (OPO) or amplification (OPA) and as efficient photonic devices [2–8]. The SHG is an NLO phenomenon which is related to the nonlinear electrical susceptibility. Developing highly efficient NLO crystals for visible, ultra-violet (UV) and deep-UV applications is important for laser spectroscopy and laser processing, including laser-tailoring of molecules and optical triggering [9–13].

The ternary semiconductor compounds AB_mC_n (A = Cu, Ag, Zn, Cd, ...; B = Al, Ga, In, La, ... and C = S, Se, Te with chalcopyrite structure have attracted considerable attention in recent years due to their applications in optical sensing, light-emitting diodes, biological labeling, solar cells, magnetic and as photocatalysts [14–24]. The non-centro-symmetric ternary semiconductor chalcogenide, CdLa₂S₄ and CdLa₂Se₄ have a strong absorption in the visible light region and, therefore, they become good nonlinear optical (NLO) crystals which can produce efficient second harmonic generation (SHG) in the visible light region. Tremendous research work has

been done on the photocatalytic performance of these materials [14–24]. However, to date, no comprehensive work neither experimental data or first principles calculations on the structural, electronic, linear and nonlinear optical properties of CdLa₂X₄ (X = S or Se) have appeared in the literature. Therefore, as a natural extension to previous existing works, the detailed depiction of the electronic structure, linear and nonlinear optical properties of CdLa₂X₄ (X = S or Se) using full potential method is timely and would bring us important insights in understanding the origin of the linear and nonlinear optical properties. Hence, it is very important to use a full potential method based on the density functional theory (DFT). The full-potential method [25] within different types of exchange correlation (XC) potentials, namely general gradient approximation (PBE-GGA) [26] and the modified Becke-Johnson potential (mBJ) [27], are used to ascertain the influence of the XC on the resulting band gap, and hence, on the linear and nonlinear optical properties of CdLa₂X₄ (X = S or Se). In this work, *ab-initio* calculations are performed to investigate the linear and nonlinear optical properties of the non-centro-symmetric CdLa₂X₄ (X = S or Se) for using as efficient NLO materials in the visible spectral region.

2. Methodology aspect

Two ternary semiconductor compounds with non-centro-symmetric chalcopyrite structure are used to investigate their

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linear and nonlinear optical properties. These are CdLa₂S₄ and CdLa₂Se₄ with tetragonal symmetry ($I\bar{4}2d$) and lattice parameters $\mathbf{a} = \mathbf{b} = 8.5919900 \text{ \AA}$, $\mathbf{c} = 9.00872271 \text{ \AA}$ and $Z = 4$ for CdLa₂S₄ [28], and $\mathbf{a} = \mathbf{b} = 8.93262736 \text{ \AA}$, $\mathbf{c} = 9.40390262 \text{ \AA}$ and $Z = 4$ for CdLa₂Se₄ [29]. The crystal structure of CdLa₂S₄ and CdLa₂Se₄ are formed through a combination of the covalent CdX₄ and LaX₄ structural units (Fig. 1). The x-ray structural data [28,29] are used as input to calculate the electronic band structure and, hence, the linear and nonlinear optical properties of CdLa₂S₄ and CdLa₂Se₄. The geometrical structures are optimized using the full-potential method (wien2k package [25]) within the generalized gradient approximation (PBE-GGA) [26]. The resulting relaxed geometrical structures are used to calculate the ground state properties utilizing the recently modified Becke-Johnson potential (mBJ) [27]. To reach accurate results, the muffin-tin spheres (R_{MT}) are chosen in such a way that the spheres did not overlap. Therefore, to insure that no charge leakage is left out of the atomic sphere cores, the R_{MT} s are chosen to be 2.5 atomic unit (a.u.) for Cd, La, and Se while it is 2.31 a.u. for S. We should emphasize that, to achieve the total energy convergence, the basis functions in the interstitial region (IR) are expanded up to $R_{\text{MT}} \times K_{\text{max}} = 7.0$ and inside the atomic spheres for the wave function, the l_{max} is taken to be equal 10, and the charge density is Fourier expanded up to $G_{\text{max}} = 12(\text{a.u.})^{-1}$. A mesh of 1000 \vec{k} points in the irreducible Brillouin zone (IBZ) is used to obtain the self-consistency which is converged since the total energy of the system is stable within 0.00001 Ry. The ground state properties, and hence the linear and nonlinear optical properties are obtained using 50000 \vec{k} points in the IBZ. The input required for linear and nonlinear optical properties calculation are the energy eigenvalues and eigenfunctions which are the natural outputs of the band structure calculation.

In recent years, due to the improvement of the computational technologies, it has been proven that the first-principles calculation is a strong and useful tool to predict the crystal structure and properties related to the electron configuration of a material before its synthesis [30–38]. It is well known that the DFT approaches have the ability to accurately predict the ground state properties of the materials, and the developed analysis tools are vital to investigating their intrinsic mechanism. This microscopic understanding has further guided molecular engineering design for new crystals with novel structures and properties. It is anticipated that first-principle material approaches highly improve the search efficiency and greatly help scientists to save resources in the exploration of new materials with good performance [30–38].

We would like to mention that, in our previous works [39–43], we have calculated the linear and nonlinear optical properties and the energy band gaps using full potential method for several systems whose linear and nonlinear optical properties and energy band gaps are known experimentally and a very good agreement with the experimental data was obtained. Thus, we believe that our

calculations reported in this paper would produce very accurate and reliable results which will greatly help to develop the experiments and save resources in the exploration of new NLO crystals with appropriate linear and nonlinear optical performance. The present study is aimed at the NLO properties of the non-centrosymmetric CdLa₂S₄ and CdLa₂Se₄ crystals.

The linear optical properties are calculated using the optical code implemented in the wien2k package [25], for more details see the user guide [44] and ref. 45. The nonlinear optical properties are calculated using the NLO code, which is compatible with the Wien2k package, see refs. 46,47. CdLa₂S₄ and CdLa₂Se₄ crystallizes in the non-centro-symmetric with tetragonal symmetry ($I\bar{4}2d$). This symmetry allows three non-zero components of the second-order dielectric (optical) tensor corresponding to the electric field \vec{E} being directed along \mathbf{a} , \mathbf{b} , and \mathbf{c} -crystallographic axes. We identify these with the x , y and z Cartesian directions.

These are $\epsilon^{xx}(\omega) = \epsilon^{yy}(\omega) = \epsilon^{\perp}(\omega)$ and $\epsilon^{zz}(\omega) = \epsilon^{\parallel}(\omega)$. The imaginary part of the principal complex tensor components completely defines the linear optical susceptibilities. The imaginary part of the principal complex tensor components originates from inter-band transitions between valence and conduction band states. According to the dipole selection rule, only transitions changing the angular momentum quantum number l by unity are allowed. The imaginary parts of the optical function's dispersion were calculated using the following expression which is taken from Ref. 44:

$$\epsilon_{2}^{ij}(\omega) = \frac{8\pi^2 \hbar^2 e^2}{m^2 V} \sum_k \sum_{cv} (f_c - f_v) \frac{p_{cv}^i(k) p_{vc}^j(k)}{E_{vc}^2} \delta[E_c(k) - E_v(k) - \hbar\omega] \quad (1)$$

where m , e and \hbar are the electron mass, charge and Planck's constant, respectively. f_c and f_v represent the Fermi distributions of the conduction and valence bands, respectively. The term $p_{cv}^i(k)$ denotes the momentum matrix element transition from the energy level c of the conduction band to the level v of the valence band at certain \mathbf{k} -point in the BZ and V is the unit cell volume.

The complex second-order nonlinear optical susceptibility tensor $\chi_{ijk}^{(2)}(-2\omega; \omega; \omega)$ can be generally written as [46–49]:

$$\chi_{inter}^{ijk}(-2\omega; \omega; \omega) = \frac{e^3}{\hbar^2} \sum_{nml} \int \frac{d\vec{k}}{4\pi^3} \frac{\vec{r}_{nm}^i \{ \vec{r}_{ml}^j \vec{r}_{ln}^k \}}{(\omega_{ln} - \omega_{ml})} \left\{ \frac{2f_{nm}}{(\omega_{mn} - 2\omega)} + \frac{f_{ml}}{(\omega_{ml} - \omega)} + \frac{f_{ln}}{(\omega_{ln} - \omega)} \right\} \quad (2)$$

$$\chi_{intra}^{ijk}(-2\omega; \omega; \omega) = \frac{e^3}{\hbar^2} \int \frac{d\vec{k}}{4\pi^3} \left[\sum_{nml} \omega_{nm} \vec{r}_{nm}^i \{ \vec{r}_{ml}^j \vec{r}_{ln}^k \} \left\{ \frac{f_{nl}}{\omega_{ln}^2(\omega_{ln} - \omega)} - \frac{f_{lm}}{\omega_{ml}^2(\omega_{ml} - \omega)} \right\} - 8i \sum_{nm} \frac{f_{nm} \vec{r}_{nm}^i \{ \Delta_{mn}^j \vec{r}_{nm}^k \}}{\omega_{mn}^2(\omega_{mn} - 2\omega)} + 2 \sum_{nml} \frac{f_{nm} \vec{r}_{nm}^i \{ \vec{r}_{ml}^j \vec{r}_{ln}^k \}}{\omega_{mn}^2(\omega_{mn} - 2\omega)} \right] \quad (3)$$

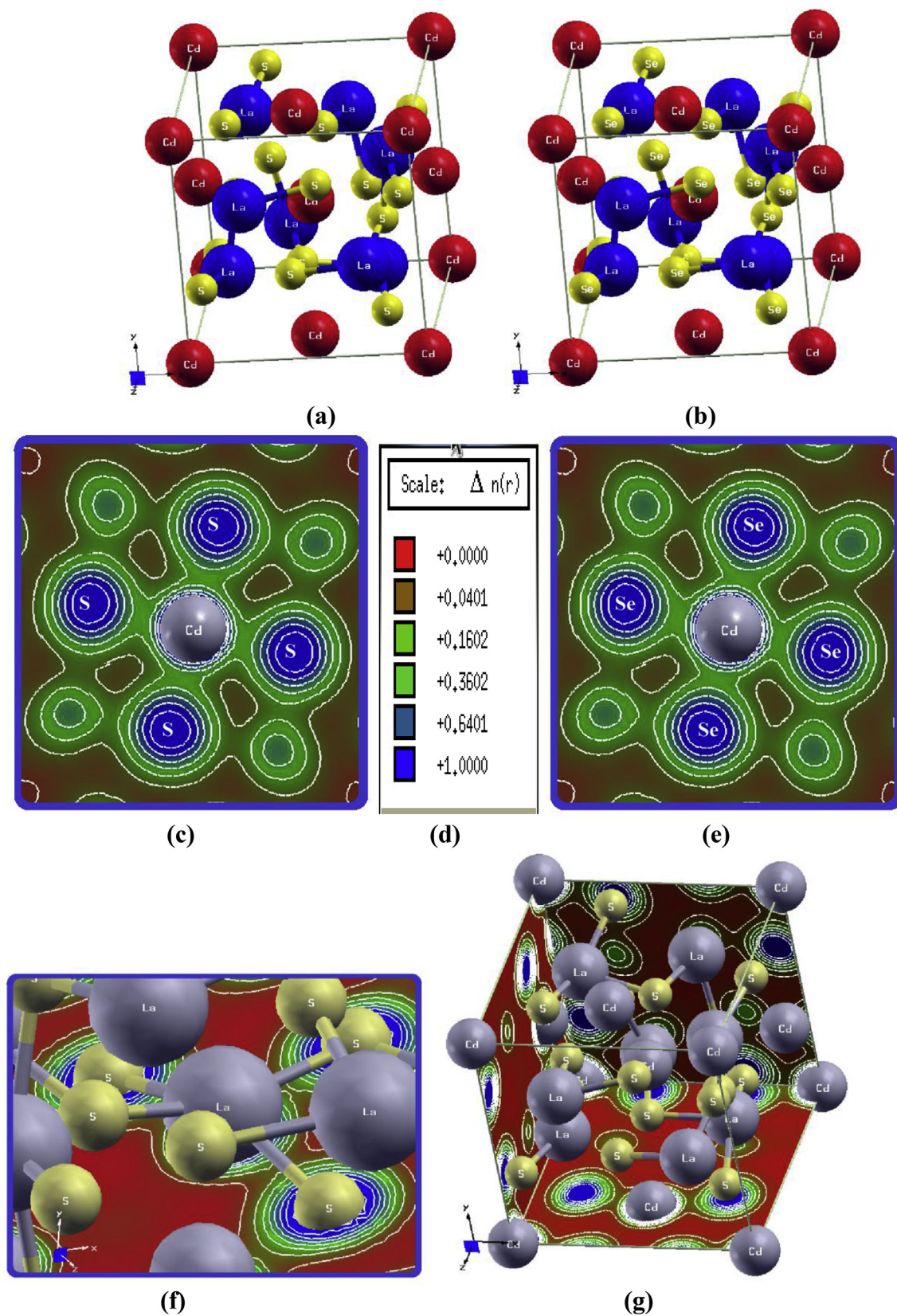


Fig. 1. (a–b) Crystal structure of CdLa_2S_4 and CdLa_2Se_4 with tetragonal symmetry ($I\bar{4}2d$) and lattice parameters $a = b = 8.5919900 \text{ \AA}$, $c = 9.00872271 \text{ \AA}$ and $Z = 4$ for CdLa_2S_4 , and $a = b = 8.93262736 \text{ \AA}$, $c = 9.40390262 \text{ \AA}$ and $Z = 4$ for CdLa_2Se_4 . The crystal structure of CdLa_2S_4 and CdLa_2Se_4 forms through a combination of the covalent CdX_4 and LaX_4 structural units; (c–g) The charge density distribution of CdLa_2S_4 and CdLa_2Se_4 which shows an efficient charge transfer occurs towards Cd, S or Se atoms, as the Cd, S or Se atoms are surrounded by uniform spheres of charge density and the maximum charge accumulates around Cd, S or Se atoms as indicated by the blue color, the thermoscale show the blue color indicates the maximum charge intensity (1.0000). (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

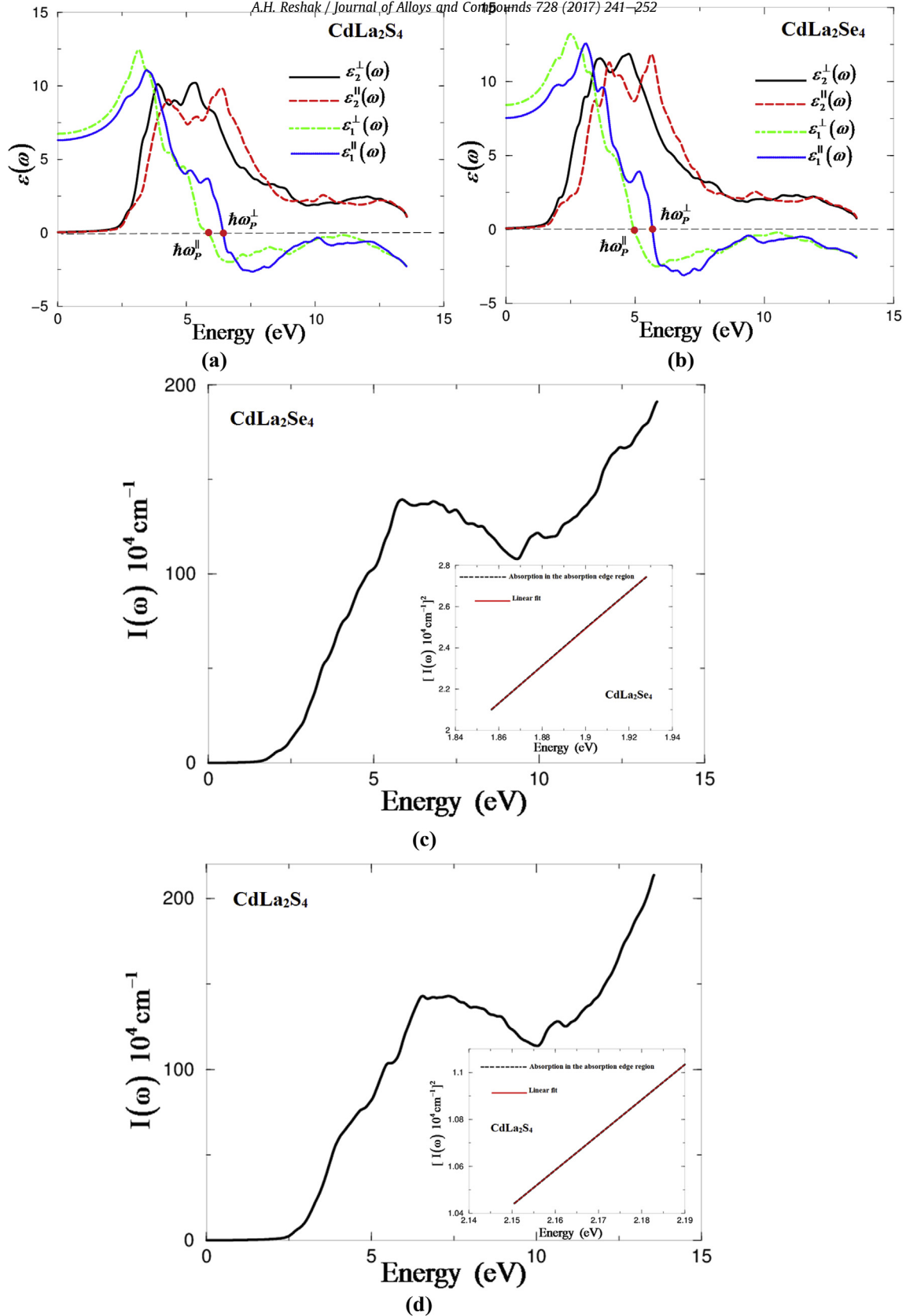


Fig. 2. (a) Calculated $\epsilon_2^\perp(\omega)$ (dark solid curve-black color online), $\epsilon_2^\parallel(\omega)$ (light long dashed curve-red color online) along with Calculated $\epsilon_1^\perp(\omega)$ (dashed dotted curve-green color online), $\epsilon_1^\parallel(\omega)$ (dotted curve-blue color online) for CdLa_2S_4 ; (b) Calculated $\epsilon_2^\perp(\omega)$ (dark solid curve-black color online), $\epsilon_2^\parallel(\omega)$ (light long dashed curve-red color online) along with Calculated $\epsilon_1^\perp(\omega)$ (dashed dotted curve-green color online), $\epsilon_1^\parallel(\omega)$ (dotted curve-blue color online) for CdLa_2Se_4 ; (c, d) Calculated absorption coefficient of CdLa_2S_4 and CdLa_2Se_4 it is clear that the direct band gap semiconductors have large absorption coefficient ($10^4 - 10^5 \text{ cm}^{-1}$). The optical band gap's value of the semiconductor materials could be solved as follow; the square of absorption coefficient $I(\omega)$ is linear with energy (E) for direct optical transitions in the absorption edge region, whereas the square root of $I(\omega)$ is linear with E for indirect optical transitions. Since the calculated electronic band structure of CdLa_2S_4 and CdLa_2Se_4 confirms the direct nature of the band gap therefore, the data plots of the square $I(\omega)$ versus E . It is clearly show that the square $I(\omega)$ versus E is linear in the absorption edge region. These plots suggest that the absorption edge of CdLa_2S_4 and CdLa_2Se_4 is caused by direct transitions. We can conclude that the absorption edges of CdLa_2S_4 and CdLa_2Se_4 occurs at $\lambda = 579.3 \text{ nm}$, and $\lambda = 670.1 \text{ nm}$, and the optical band gaps are estimated

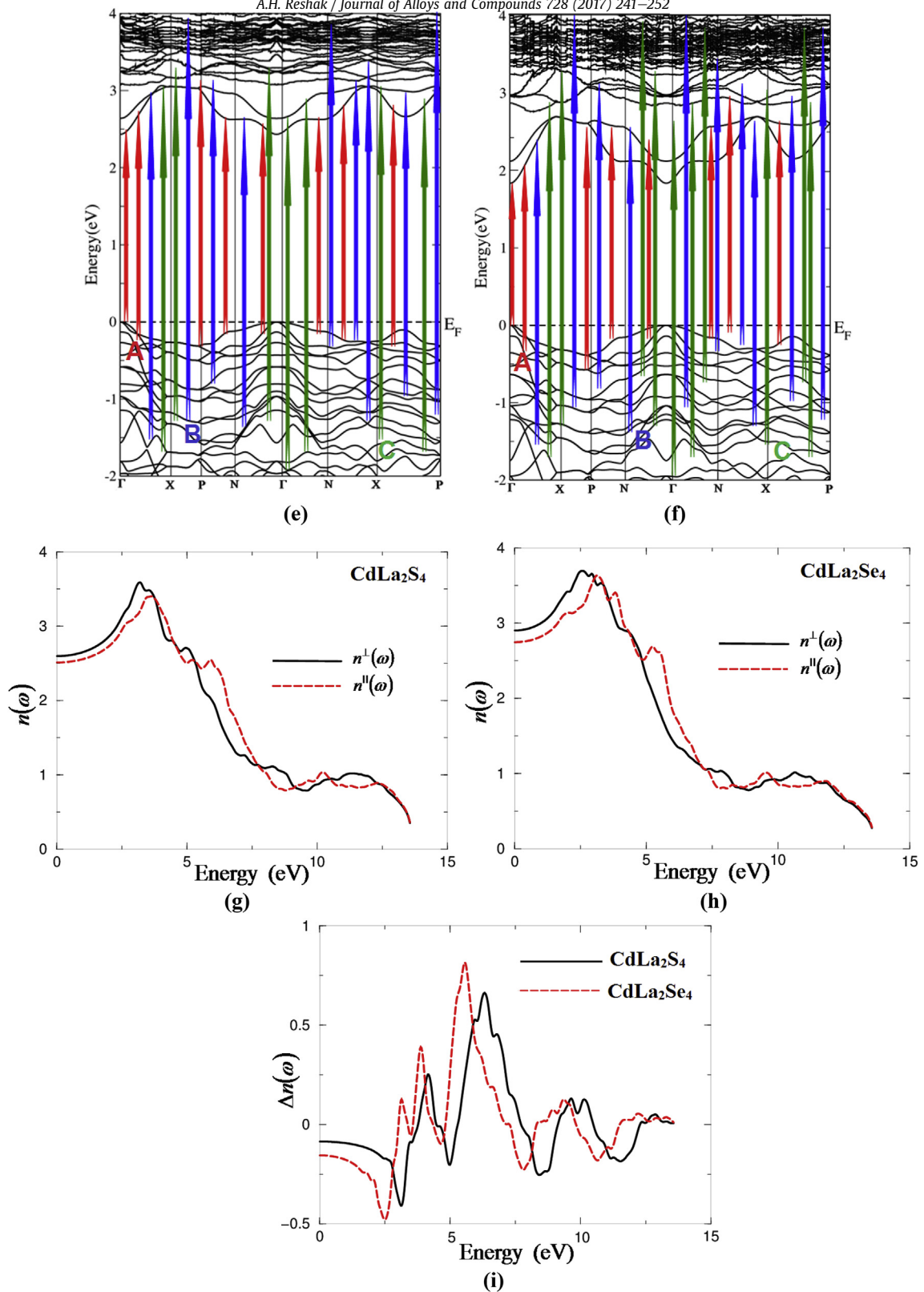


Fig. 2. (continued).

to be 2.14 eV, and 1.85 eV for CdLa₂S₄ and CdLa₂Se₄, respectively; (e, f) The optical transitions depicted on a generic band structure of CdLa₂S₄ and CdLa₂Se₄; (h, g) Calculated $n^{\perp}(\omega)$ (dark solid curve-black color online), $n^{\parallel}(\omega)$ (light long dashed curve-red color online) for CdLa₂S₄ and CdLa₂Se₄; (i) Calculated $\Delta n(\omega)$ (dark solid curve-black color online) for CdLa₂S₄ and (light long dashed curve-red color online) for CdLa₂Se₄; (j, k) Calculated $R^{\perp}(\omega)$ (dark solid curve-black color online), $R^{\parallel}(\omega)$ (light long dashed curve-red color online) for CdLa₂S₄ and CdLa₂Se₄; (l, m) Calculated $\sigma_2^{\perp}(\omega)$ (dark solid curve-black color online), $\sigma_2^{\parallel}(\omega)$ (light long dashed curve-red color online) along with Calculated σ_1^+ (dashed dotted curve-green color online), σ_1^- (dotted curve-blue color online) for CdLa₂S₄ and CdLa₂Se₄; (n, o) Calculated $L^{\perp}(\omega)$ (dark solid curve-black color online), $L^{\parallel}(\omega)$ (light long dashed curve-red color online) for CdLa₂S₄ and CdLa₂Se₄. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

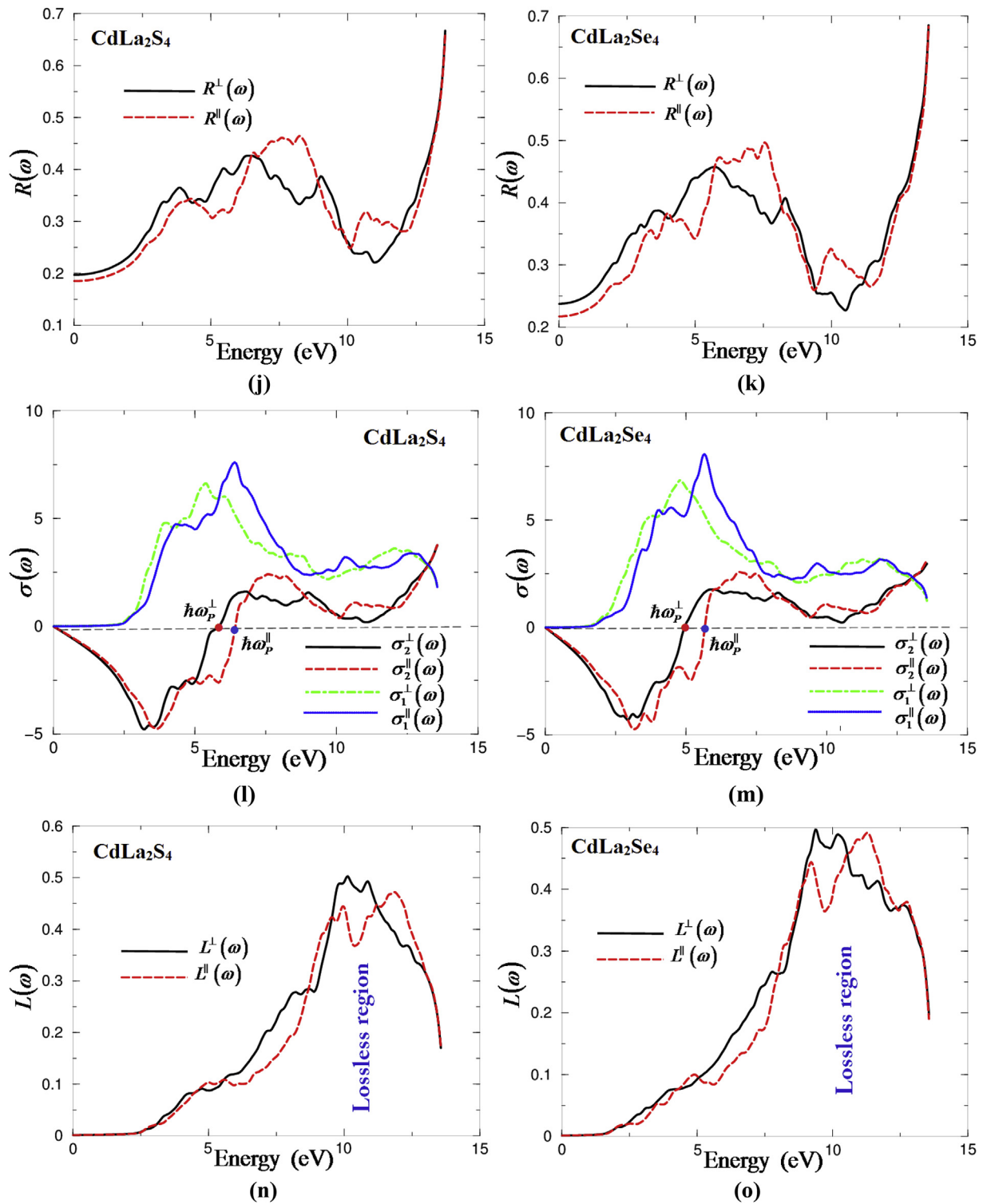


Fig. 2. (continued).

$$\chi_{\text{mod}}^{ijk}(-2\omega; \omega, \omega) = \frac{e^3}{2\hbar^2} \int \frac{d\vec{k}}{4\pi^3} \left[\sum_{nml} \frac{f_{nm}}{\omega_{nm}^2(\omega_{nm} - \omega)} \left\{ \omega_{nl} \vec{r}_{lm}^i \left\{ \vec{r}_{mn}^j \vec{r}_{nl}^k \right\} - \omega_{lm} \vec{r}_{nl}^i \left\{ \vec{r}_{lm}^j \vec{r}_{mn}^k \right\} \right\} - i \sum_{nm} \frac{f_{nm} \vec{r}_{nm}^i \left\{ \vec{r}_{mn}^j \Delta_{mn}^k \right\}}{\omega_{nm}^2(\omega_{nm} - \omega)} \right] \quad (4)$$

From these formulae we can notice that there are three major contributions to $\chi_{ijk}^{(2)}(-2\omega; \omega, \omega)$: the inter-band transitions $\chi_{\text{inter}}^{ijk}(-2\omega; \omega, \omega)$, the intra-band transitions $\chi_{\text{intra}}^{ijk}(-2\omega; \omega, \omega)$ and the modulation of inter-band terms by intra-band terms $\chi_{\text{mod}}^{ijk}(-2\omega; \omega, \omega)$, where $n \neq m \neq l$. Here n denotes the valence states, m the conduction states and l denotes all states ($l \neq m, n$). There are two kinds of transitions that can take place, one of them vcc' , involves one valence band (v) and two conduction bands (c and c'), and the second transition $vv'c$, involves two valence bands (v and v') and one conduction band (c). The symbols are defined as $\Delta_{nm}^i(\vec{k}) = \vartheta_{nm}^i(\vec{k}) - \vartheta_{mm}^i(\vec{k})$ with ϑ_{nm}^i being the i component of the electron velocity given as $\vartheta_{nm}^i(\vec{k}) = i\omega_{nm}(\vec{k})r_{nm}^i(\vec{k})$ and $\{r_{nm}^i(\vec{k})r_{ml}^j(\vec{k})\} = \frac{1}{2}(r_{nm}^i(\vec{k})r_{ml}^j(\vec{k}) + r_{nm}^j(\vec{k})r_{ml}^i(\vec{k}))$. The position matrix elements between band states n and m , $r_{nm}^i(\vec{k})$, are calculated from the momentum matrix element P_{nm}^i using the relation [44]: $r_{nm}^i(\vec{k}) = \frac{P_{nm}^i(\vec{k})}{im\omega_{nm}(\vec{k})}$, with the energy difference between the states n and m given by $\hbar\omega_{nm} = \hbar(\omega_n - \omega_m)$. $f_{nm} = f_n - f_m$ is the difference of the Fermi distribution functions. i, j and k correspond to cartesian indices. It has been demonstrated by Aspnes [50] that only the one-electron virtual transitions (transitions between one valence band state and two conduction band states vcc') give a significant contribution to the second-order tensor. We ignore the virtual-hole contribution (transitions between two valence band states and one conduction band state, $vv'c$) because it was found to be negative and more than an order of magnitude smaller than the virtual-electron contribution for the considered compounds. For simplicity, we denote $\chi_{ijk}^{(2)}(-2\omega; \omega, \omega)$ by Refs. $\chi_{ijk}^{(2)}(\omega)$ [46–50]. The subscripts i, j , and k are cartesian indices.

3. Results and discussion

3.1. Complex first-order linear optical dispersion

Analyzing the electronic distribution is an important factor to investigate the origin of optical properties [45,51]. Deep insight into the electronic structure can be obtained from analyzing the electron cloud of the CdX_4 and LaX_4 units (Fig. 1 c–g). The electron cloud of the CdX_4 groups exhibits a planar shape with conjugated electron orbitals (Fig. 1(e)) shows a high electron density configuration and a strong anisotropy for Cd-X bonds, resulting in the optical anisotropy of CdX_4 groups. The charge localizes mainly between Cd and the neighboring S or Se atoms in CdX_4 unit, also between La and S or Se atoms in LaX_4 unit indicating a partial ionic and strong covalent bonding. The strength of the interactions between the atoms is due to the degree of the hybridization and the electronegativity differences. Also, due to the electronegativity differences between the atoms, some valence electrons are

transferred to S or Se atoms and these atoms are surrounded by uniform blue spheres which indicate the maximum charge accumulation according to thermoscale (Fig. 1(d)). The Cd–S, Cd–Se, La–S and La–Se units possess strong electron cloud overlap and prefer to attract holes and repel electrons this in turn enhances the optical activity. The figures bring a clear visualization helpful to understand the origin of the optical properties of these two compounds. The covalent bonding is more favorable for the carrier transport than the ionic one [52]. This may be one of the reasons for the strong linear and nonlinear optical properties in CdLa_2S_4 and CdLa_2Se_4 . The electron cloud of the CdX_4 groups exhibits a planar shape with conjugated electron orbitals. Thus, CdX_4 can be the source of the large birefringence in CdLa_2S_4 and CdLa_2Se_4 . It is well-known that the birefringence determines partly whether an NLO material has the value of study [53–56]. Furthermore, Fig. 1(c–g) show a high electron density configuration and strong anisotropy for Cd-X groups, which indicates the main contribution of CdX_4 groups to the optical anisotropy.

The optical properties can provide more detailed information about the electronic structure of the materials. It is well-known that the optical properties are very sensitive to the energy band gap. In the density functional theory (DFT), by solving the Kohn-Sham equations, we map an interacting many-body system to a non-interacting hypothetical system which has the same electron density. The price that we will pay is the definition of a new functional that is called the exchange-correlation functional. Unfortunately, the exact form of exchange-correlation functional is unknown. Therefore, the accuracy of our results will be sensitive to selection of the exchange-correlation functional and it can play a major role in the accuracy of our results, and this is one of the main questions in DFT. Thus, based on our previous experiences with using mBJ, we have used mBJ to calculate the electronic band structures and hence the optical properties of CdLa_2S_4 and CdLa_2Se_4 [57,58]. The imaginary part of the linear optical properties $\epsilon_2^{xx}(\omega)$, $\epsilon_2^{yy}(\omega)$ and $\epsilon_2^{zz}(\omega)$ for the tetragonal system (CdLa_2S_4 and CdLa_2Se_4) are calculated and presented in Fig. 2(a and b). It has been noticed that the substitution of S by Se leads to a shift of the whole spectral structure towards lower energies and increase the magnitudes of the optical components.

The spectral structure of CdLa_2S_4 and CdLa_2Se_4 exhibits two main peaks for $\epsilon_2^x(\omega)$ and $\epsilon_2^y(\omega)$ components with small humps situated on the left and right shoulders. The absorption edges occur at 2.14 eV (CdLa_2S_4) and 1.85 eV (CdLa_2Se_4), as shown in Fig. 2(a and b) and it is confirmed by the calculated absorption coefficients shown in Fig. 2(c and d), which show these compounds possess large absorption coefficient (10^4 cm^{-1}). The optical band gap is crucial for elucidation of the linear and nonlinear optical properties since the energy band gap value appears in the denominator of the equations used to calculate the linear and nonlinear optical properties [eqs. (1)–(4)]. The optical band gap value of the semiconductor materials could be solved as follow; the square of absorption coefficient $I(\omega)$ is linear with energy (E) for direct optical

transitions in the absorption edge region, whereas the square root of $I(\omega)$ is linear with E for indirect optical transitions [59,60]. Since the calculated electronic band structure of CdLa₂S₄ and CdLa₂Se₄ confirms the direct nature of the band gap, the data plots of the square $I(\omega)$ versus E are shown in Fig. 2 (c, d). It is clearly show that the square $I(\omega)$ versus E is linear in the absorption edge region. These plots suggest that the absorption edges of CdLa₂S₄ and CdLa₂Se₄ are caused by direct transitions. Following Fig. 2 (c, d), we can conclude that the absorption edges of CdLa₂S₄ and CdLa₂Se₄ occur at $\lambda = 579.3$ nm, and $\lambda = 670.1$ nm. The optical band gaps are estimated ($\lambda_g = 1239.8/E_{g(optical)}$) [61] to be 2.14 eV, and 1.85 eV for CdLa₂S₄ and CdLa₂Se₄, respectively. The obtained band gap of CdLa₂S₄ show good agreement with the available experimental data (2.39 eV) [18,62] and much better than the reported values 1.785 eV (CdLa₂S₄) and 1.099 eV (CdLa₂Se₄) using VASP code within GGA [28,29], due to the fact that GGA use to underestimated the energy band gap value [63]. The mBJ is a modified Becke-Johnson potential, which allows the calculation of the energy band gap with accuracy similar to that of the very expensive GW calculations [27]. It is a local approximation to an atomic “exact-exchange” potential and a screening term.

The absorption edges of CdLa₂S₄ (CdLa₂Se₄) are originated from the optical transitions between S-p, La-d (Se-p, La-d) and Cd-s, S-p, La-d (Cd-s, Se-p, La-d) states. It is clear that the Cd orbitals exhibit an insignificant contribution to CBM and VBM; thus, variations in divalent elements can be systematically explored to assess the effects on SHG efficiencies, as well as other attractive properties, while maintaining a wide band gap [64]. The optical transitions are calculated from the momentum matrix elements between the occupied and unoccupied bands, giving rise to the selection rules, as shown in Fig. 2(e and f). In order to identify the observed spectral structures we need to look at the magnitude of the optical matrix elements. The observed spectral structures would correspond to those transitions that have large optical matrix elements. We have used our calculated electronic band structure to map the allowed optical transitions following the selection rules. For simplicity, we have labeled the optical transitions as A, B, and C. The A transitions are responsible for the optical spectral structures between (0.0–5.0) eV, the B transitions represent (5.0–10.0) eV and the C transitions represent the spectral structures between (10.0–14.0) eV (see Fig. 2(e and f)).

Furthermore, the calculated real part of the optical dielectric function can give information about the energy gap since the calculated static electronic dielectric constant by $\epsilon_\infty = \epsilon_1(0)$ is inversely related to the energy gap, this could be explained on the basis of the Penn model [65]. As it is clear that the values of $\epsilon_1^\perp(0)$ and $\epsilon_1^\parallel(0)$ for CdLa₂S₄ are lower than those obtained for CdLa₂Se₄, hence higher $\epsilon_1(0)$ exhibits a lower energy gap. This supports our finding that substituting S by Se cause a band gap reduction. The calculated values of $\epsilon_1^\perp(0)$ and $\epsilon_1^\parallel(0)$ along with the calculated plasmon oscillations ω_p^\perp and ω_p^\parallel for both compounds are presented in Table 1. The plasmon oscillations are associated with inter-band transitions that occur at energy where optical spectra of the real part crosses zero. The optical components exhibit a considerable anisotropy; this is one of the important features of the optical spectra. The other important feature is the uniaxial anisotropy ($\delta\epsilon$) which can be obtained from $\epsilon_1^\perp(0)$ and $\epsilon_1^\parallel(0)$. Following Fig. 2(a and b), we found that both compounds possess negative $\delta\epsilon$, as shown in Table 1.

The calculated refractive indices $n^\perp(\omega)$ and $n^\parallel(\omega)$ of CdLa₂S₄ and CdLa₂Se₄ as derived from the complex dielectric functions are presented in Fig. 2(g and h). Thus, the birefringence can be estimated from $\Delta n(\omega) = n_e(\omega) - n_o(\omega)$. It was found that both

compounds exhibit negative birefringence at the static limit, as shown in Table 1. We found that the $\Delta n(0)$ for CdLa₂S₄ > CdLa₂Se₄; hence CdLa₂Se₄ exhibits weaker anisotropy. The birefringence is necessary to fulfill the phase-matching condition. Therefore, we expected CdLa₂Se₄ will produce higher second harmonic generation than CdLa₂S₄. The dispersion of the birefringence of the two compounds is illustrated in Fig. 2(i).

The optical reflectivity spectra of CdLa₂S₄ and CdLa₂Se₄ along the [100], [101] and [001] polarization directions are shown in Fig. 2(j and k). It was been found that at low energies the compounds exhibit low reflectivity. The first reflectivity maxima occur at around 7.0 eV, followed by the first reflectivity minima which is situated at around 11.0 eV, confirming the occurrence of a collective plasmon resonance in concordance with our observation in Fig. 2(a and b). At higher energies the region confined between 12.0 and 12.0 eV represents the lossless region.

The optical conductivity (Fig. 2 l,m) can be obtained from the complex first-order linear optical dielectric function following the expression $\epsilon(\omega) = \epsilon_1(\omega) + i\epsilon_2(\omega) = 1 + \frac{4\pi\sigma(\omega)}{\omega}$. It consists of imaginary and real parts; therefore, it completely characterizes the linear optical properties. The imaginary part $\sigma_2^\perp(\omega)$ and $\sigma_2^\parallel(\omega)$ between 0.0 and the values of ω_p^\perp and ω_p^\parallel exhibit overturned features of $\epsilon_2^\perp(\omega)$ and $\epsilon_2^\parallel(\omega)$, whereas the real parts $\sigma_1^\perp(\omega)$ and $\sigma_1^\parallel(\omega)$ show similar features to those of $\epsilon_2^\perp(\omega)$ and $\epsilon_2^\parallel(\omega)$. The intersection of the imaginary and real parts of the optical conductivity at the zero energy represents the values of ω_p^\perp and ω_p^\parallel . The loss function peaks (Fig. 2 n,o), are initiated at the values of the plasma frequencies ω_p^\perp and ω_p^\parallel at the energy point where optical spectra of $\epsilon_1^\perp(\omega)$ and $\epsilon_1^\parallel(\omega)$ cross zero.

3.2. Complex second-order non-linear optical dispersion

Since the nonlinear optical properties are very sensitive to the energy band gap values due to the fact that the band gap value comes in the denominator of the complex nonlinear optical formulas [eqs. (2)–(4)] which are used to calculate the complex nonlinear optical susceptibility. Therefore, to achieve appropriate results, an accurate energy band gap is required, and, to achieve this, the mBJ approximation was applied. Furthermore, a quasi-particle self-energy correction at the level of scissors operators is applied to avoid the DFT drawback. In the scissors operators, the energy bands are rigidly shifted to merely bring the DFT band gap to the exact experimental value. The tetragonal symmetry (*I*-42*d*) allows only two non-zero tensor components. These are

Table 1

The calculated energy band gap in comparison with the experimental value, $\epsilon_1^\perp(0)$, $\epsilon_1^\parallel(0)$, $\delta\epsilon$, ω_p^\perp , ω_p^\parallel , $n^\perp(0)$, $n^\parallel(0)$ and $\Delta n(0)$.

	CdLa ₂ S ₄		CdLa ₂ Se ₄	
	This work	other	This work	other
E _g (eV)	2.14	2.39 ^a , 1.785 ^b	1.85	1.099 ^c
$\epsilon_1^\perp(0)$	6.749		8.424	
$\epsilon_1^\parallel(0)$	6.311		7.546	
$\delta\epsilon$	−0.070		−0.109	
ω_p^\perp	5.864		4.966	
ω_p^\parallel	6.435		5.673	
$n^\perp(0)$	2.598		2.902	
$n^\parallel(0)$	2.512		2.747	
$\Delta n(0)$	−0.086		−0.155	

^a Refs. [18, 62].

^b Ref. [28].

^c Ref. [29].

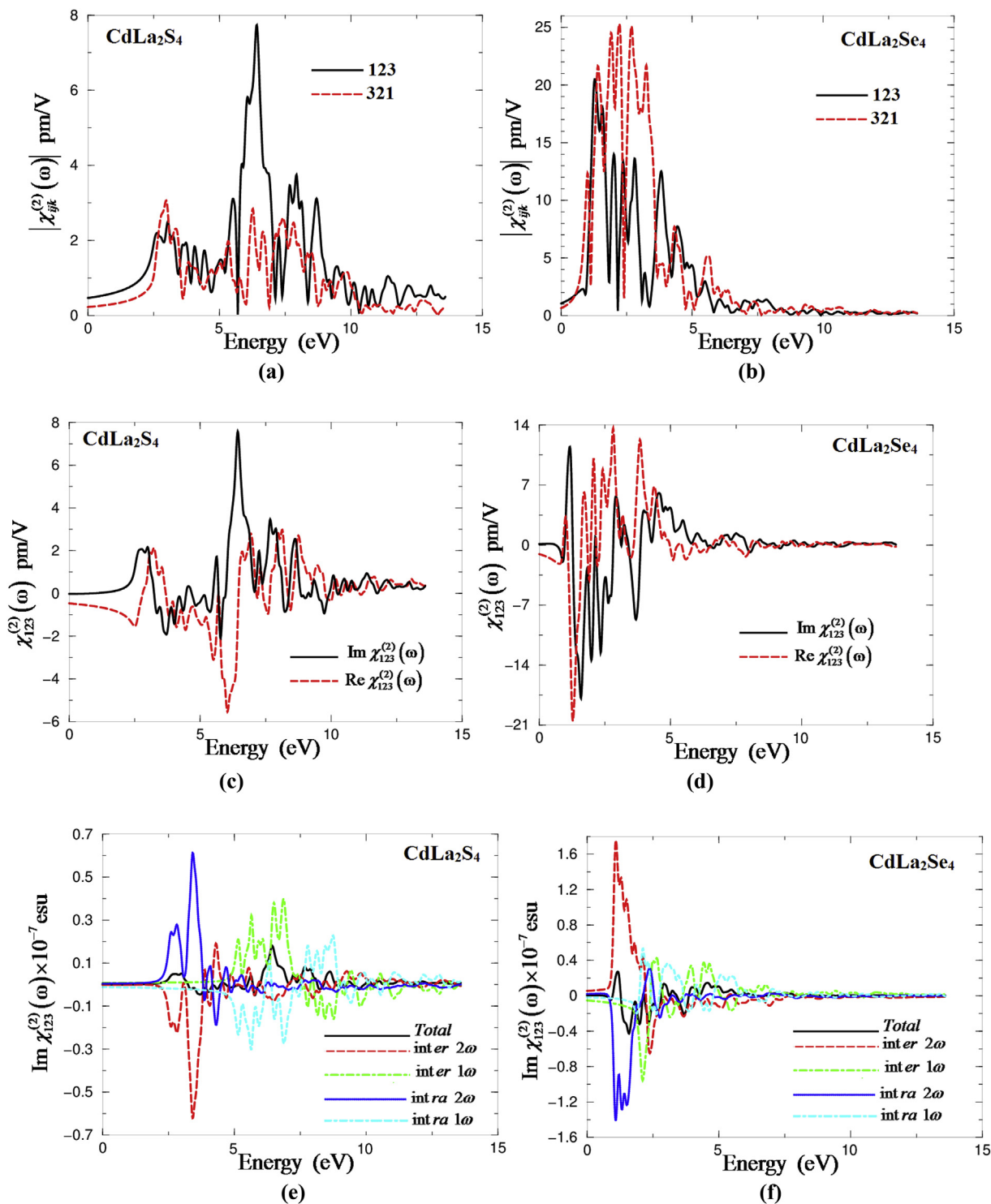


Fig. 3. (a) Calculated $|\chi_{ijk}^{(2)}(\omega)|$ for the two tensor components of CdLa₂S₄; (b) Calculated $|\chi_{ijk}^{(2)}(\omega)|$ for the two tensor components of CdLa₂Se₄; (c) Calculated Imaginary $\chi_{123}^{(2)}(\omega)$ (dark solid curve-black color online) and real $\chi_{123}^{(2)}(\omega)$ (light dashed curve-red color online) spectra of CdLa₂S₄; (d) Calculated Imaginary $\chi_{123}^{(2)}(\omega)$ (dark solid curve-black color online) and real $\chi_{123}^{(2)}(\omega)$ (light dashed curve-red color online) spectra of CdLa₂Se₄; (e, f) Calculated total $\text{Im}\chi_{123}^{(2)}(\omega)$ spectrum (dark solid curve-black color online) along with the intra $(2\omega)/(1\omega)$ (light solid curve-blue color online)/(light dashed dotted curve-cyan color online) and inter $(2\omega)/(1\omega)$ (light long dashed curve-red color online)/(light dotted curve-green color online) -band contributions of CdLa₂S₄ and CdLa₂Se₄, here all $\text{Im}\chi_{123}^{(2)}(\omega)$ are multiplied by 10^{-7} , in esu units; (g, h) –upper panel- Calculated $|\chi_{ijk}^{(2)}(\omega)|$ (dark solid curve-black color online); -lower panel- Calculated $\varepsilon_2(\omega)$ (dark solid curve-black color online); Calculated $\varepsilon_2(\omega/2)$ (dark dashed curve-red color online) of CdLa₂S₄ and CdLa₂Se₄. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

$\chi_{123}^{(2)}(-2\omega; \omega; \omega)$ and $\chi_{321}^{(2)}(-2\omega; \omega; \omega)$. Fig. 3 (a, b) show the calculated absolute values $|\chi_{ijk}^{(2)}(-2\omega; \omega; \omega)|$ of those two tensor

components for CdLa₂S₄ and CdLa₂Se₄. These figures reveals that the $|\chi_{123}^{(2)}(-2\omega; \omega; \omega)|$ is the dominant one, since it shows higher

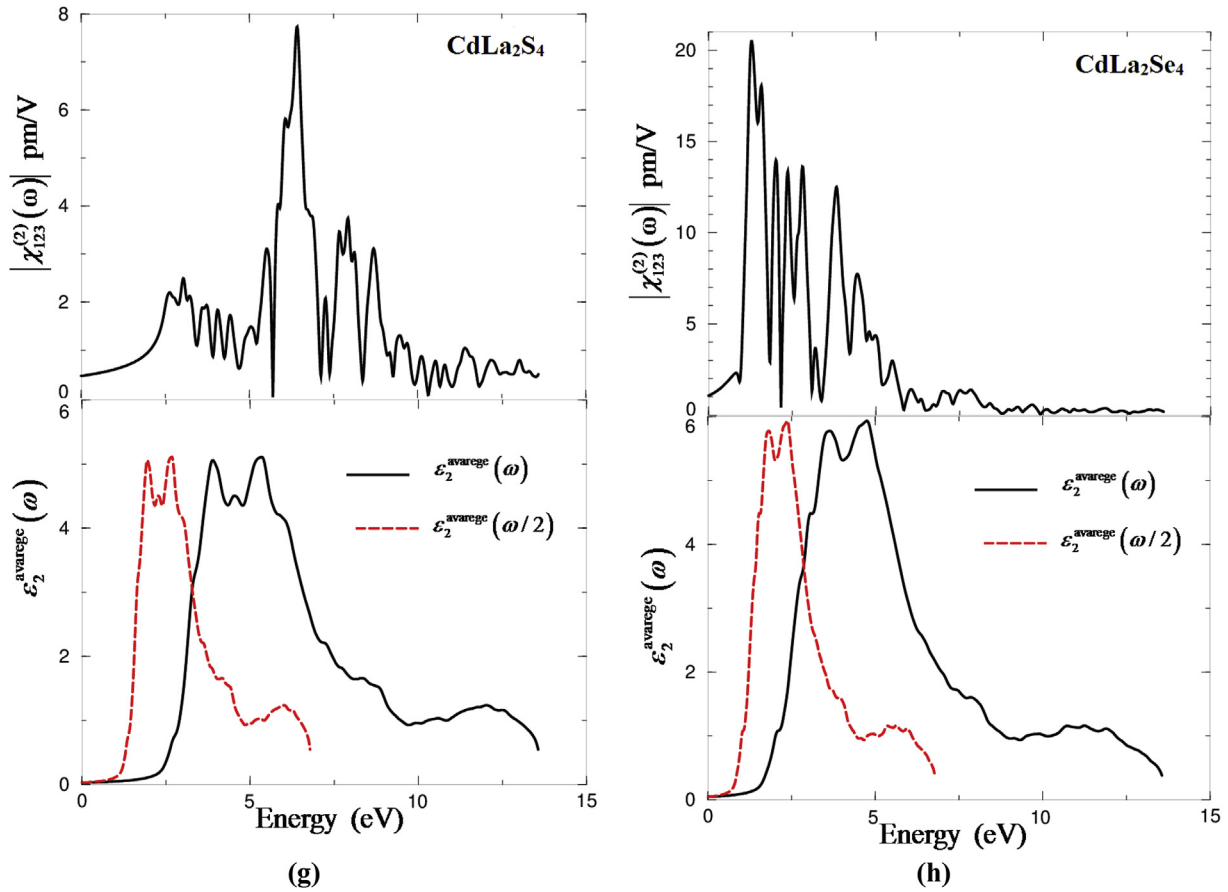


Fig. 3. (continued).

value than that of $|\chi_{321}^{(2)}(-2\omega; \omega; \omega)|$, as shown in Tables 2 and 3. Following these results, one can see that CdLa₂S₄ exhibits SHG intensity of about two times of the well-known KH₂PO₄ (KDP) crystal

which exhibits a SHG value of 0.39 p.m./V at $\lambda = 1064$ nm [66]. Whereas, CdLa₂Se₄ exhibits SHG very close to the experimental value of the well-known KTiOPO₄ (KTP) single crystals which possesses a SHG value of about 16.9 [67], 13.7 [68], 15.4 ± 0.2 [69],

Table 2
Calculated $|\chi_{ijk}^{(2)}(\omega)|$ and β_{ijk} of CdLa₂S₄, in pm/V at static limit and at $\lambda = 1064$ nm, in comparison with the experimental value of the well known KH₂PO₄ (KDP) single crystals which exhibits a SHG value of about 0.39 p.m./V at $\lambda = 1064$ nm [66]. Where $1 \text{ p.m./V} = 2.387 \times 10^{-9}$ esu.

CdLa ₂ S ₄				
Tensor components	$\chi_{ijk}^{(2)}(0)$	Theory $d_{ijk} = 0.5\chi_{ijk}^{(2)}(\omega)$	$\chi_{ijk}^{(2)}(\omega)$ at $\lambda = 1064$	Theory $d_{ijk} = 0.5\chi_{ijk}^{(2)}(\omega)$
$ \chi_{123}^{(2)}(\omega) $	0.451	$d_{14}=0.225$	0.639	$d_{14}=0.319$
$ \chi_{321}^{(2)}(\omega) $	0.216	$d_{36}=0.108$	0.319	$d_{36}=0.159$
β_{333}	0.326×10^{-30} esu		0.453×10^{-30} esu	

Table 3
Calculated $|\chi_{ijk}^{(2)}(\omega)|$ and β_{ijk} of CdLa₂Se₄ in pm/V at static limit and at $\lambda = 1064$ nm, in comparison with the experimental value of the well known KTiOPO₄ (KTP) single crystals which exhibits a SHG value of about 16.9 [67], 13.7 [68], 15.4 ± 0.2 [69], 14.6 ± 1.0 [70], 17.4 ± 1.7 [71], 16.9 ± 3.3 [72], 16.9 ± 1.7 [73], 10.6 ± 7.5 [74], 16.75 [75] and 16.65 [75] at $\lambda = 1064$ nm. Where $1 \text{ p.m./V} = 2.387 \times 10^{-9}$ esu.

CdLa ₂ Se ₄				
Tensor components	$\chi_{ijk}^{(2)}(0)$	Theory $d_{ijk} = 0.5\chi_{ijk}^{(2)}(\omega)$	$\chi_{ijk}^{(2)}(\omega)$ at $\lambda = 1064$	Theory $d_{ijk} = 0.5\chi_{ijk}^{(2)}(\omega)$
$ \chi_{123}^{(2)}(\omega) $	1.087	$d_{14}=0.543$	14.871	$d_{14}=7.435$
$ \chi_{321}^{(2)}(\omega) $	0.678	$d_{36}=0.339$	7.011	$d_{36}=3.505$
β_{333}	0.922×10^{-30} esu		12.090×10^{-30} esu	

14.6 ± 1.0 [70], 17.4 ± 1.7 [71], 16.9 ± 3.3 [72], 16.9 ± 1.7 [73], 10.6 ± 7.5 [74], 16.75 [75] and 16.65 [75] at $\lambda = 1064$ nm. It is clear that CdLa₂Se₄ exhibits larger nonlinear optical properties than that of CdLa₂S₄, and the cost of good SHG performance is the narrower band gap, which limits the transmittance range. The main point to obtain NLO materials is getting the delicate balance between the SHG response and band gap [76].

The real and imaginary parts of $\chi_{123}^{(2)}(-2\omega; \omega; \omega)$ for CdLa₂S₄ and CdLa₂Se₄ are presented in Fig. 3 (c, d). These figures show that the Im $\chi_{123}^{(2)}(-2\omega; \omega; \omega)$ value rises at a half value of the energy band gap due to 2ω terms oscillation, then at the exact value of the fundamental gap the ω terms start to oscillate to be added to 2ω terms. While at higher energies only ω terms will contribute. To analyze the spectral features of the Im $\chi_{123}^{(2)}(-2\omega; \omega; \omega)$, we have calculated the $2\omega/\omega$ inter-/intra-band contributions to explore the origin of the strong SHG in these compounds, as shown in Fig. 3(e and f). For deep investigation, the spectral structure of the dominant component is drawn in associated with the absorptive part of the corresponding dielectric function $\varepsilon_2(\omega)$ as a function of $\omega/2$ and ω , to identify the origin of the spectral peaks as caused by $2\omega/\omega$ inter-/intra-band contributions. These are shown in Fig. 3(g and h), the first structure in $|\chi_{123}^{(2)}(\omega)|$ between 1.19 and 2.39 eV (0.93–1.85 eV) for CdLa₂S₄ (CdLa₂Se₄) is mainly originated from 2ω resonance [see $\varepsilon_2(\omega/2)$ Fig. 3(g and h) –lower panel]. The second structure between 2.39 and 5.0 eV (1.85–5.0 eV) is associated with interference between 2ω and ω resonances (the threshold of $\varepsilon_2(\omega)$) [see $\varepsilon_2(\omega/2)$ and $\varepsilon_2(\omega)$ Fig. 3(g and h) –lower panel]. The last spectral structure from 5.0 and above is mainly due to ω resonance and it is associated with the second structure in $\varepsilon_2(\omega)$. With the aid of the existing values of the SHG for the dominant components of CdLa₂S₄ and CdLa₂Se₄, we obtained the values of the microscopic first hyperpolarizability, β_{123} [77,78], the vector component along the dipole moment direction, at the static limit and at $\lambda = 1064$ nm. These values are listed in Tables 2 and 3. In general, the microscopic first hyperpolarizability term, β_{ijk} , cumulatively yield a bulk observable second order susceptibility term, $\chi_{ijk}^{(2)}(\omega)$, which in turn is responsible for the high SHG response [77,78].

4. Conclusions

The influence of replacing S by Se on the linear and nonlinear optical susceptibilities in CdLa₂S₄ and CdLa₂Se₄ was investigated by means of DFT. The all-electron full-potential linearized augmented plane wave plus local orbitals (FP-LAPW + lo) method was used. Using PBE-GGA, the experimental geometries of CdLa₂S₄ and CdLa₂Se₄ were optimized by minimizing the forces acting on the atoms. The resulting geometries were used to calculate the linear and nonlinear optical susceptibilities using mBJ potential. It was been found that mBJ succeeds by a large amount in bringing the calculated energy gap closer to the experimental one. The calculation shows the direct band nature of the investigated compounds. The valence electronic charge density exhibits a clear visualization to understand the origin of linear and nonlinear optical susceptibilities. The linear optical properties exhibit considerable anisotropy that is favorable to enhance the spectral range for frequency conversion. CdLa₂S₄ exhibits SHG of about two times of the well-known KH₂PO₄ (KDP) single crystals which exhibits a SHG value of 0.39 p.m./V. Whereas CdLa₂Se₄ exhibits SHG very close to the experimental value of the well-known KTiOPO₄ (KTP) single crystals which exhibits a SHG value of about 16.9, 13.7, 15.4 ± 0.2 , 14.6 ± 1.0 , 17.4 ± 1.7 , 16.9 ± 3.3 , 16.9 ± 1.7 , 10.6 ± 7.5 , 16.75 and 16.65 .

Author contribution

A. H. Reshak, as a professor with PhD in physics and PhD in materials engineering has performed the calculations, analyzing and discussing the results and writing the manuscript.

Competing financial interests

The author declare no competing financial interests.

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